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Computational investigation of thermophysical and structural properties of molten $\text{NaCl-PuCl}_3\text{-AmCl}_3$

September 2025

Manh-Thuong Nguyen

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Manh-Thuong Nguyen

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the U.S. Department of Energy
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Summary

This work is aimed at providing thermophysical properties of molten $\text{NaCl-PuCl}_3\text{-AmCl}_3$, which is highly relevant to molten salt reactors. First, ab initio molecular dynamics (AIMD) simulations were conducted to generate high-fidelity reference datasets, on which a machine learning potential was trained. This potential, designed to preserve AIMD-level accuracy while allowing for extensive sampling, was then used in molecular dynamics simulations to calculate various thermophysical properties of the system. Specifically, the density, thermal expansion coefficient, heat capacity, viscosity, and thermal conductivity were evaluated. Moreover, temperature-dependent relationships for the density, thermal expansion coefficient, and viscosity were established. In addition to thermophysical calculations, structural properties of the mixture were also analyzed.

Acknowledgments

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Acronyms and Abbreviations

AIMD	Ab initio molecular dynamics
C _p	Constant pressure heat capacity
DFT	Density functional theory
GPW	Gaussian plane wave
H	Enthalpy
LAMMPS	Large-scale Atomic/Molecular Massively Parallel Simulator
MLIP	Machine learning interatomic potential
MS	Molten salt
MSR	Molten salt reactor
RMSEs	Root mean square errors

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1.0 Introduction

In molten salt reactors (MSRs), minor actinides are closely related to advanced fuel cycles, contributing to both energy production and long-term waste reduction. Americium trichloride (AmCl_3), in particular, is highly relevant to chloride-based MSR systems. Significant progress has been made in advancing the fundamental understanding of AmCl_3 through multiple research avenues. At the molecular level, electronic structure theory has been used to investigate its gas-phase complexes and solid-state properties, providing insights into bonding, electronic, and structural behavior (Vetere et al. 2004; Li et al. 2023). Experimentally, the synthesis of AmCl_3 has received increasing attention, with several alternative strategies being employed (Hayashi et al. 2008; Kersten et al. 2022; Chevreux et al. 2024). In parallel, the electrochemical behavior of AmCl_3 -bearing molten salts (MSs) has been investigated (Filatov et al. 2023; Serp et al. 2006), highlighting its potential role in actinide separation, redox control, and MSR applications. All together, these studies underscore the importance of AmCl_3 across theory, synthesis, and applied electrochemistry, while also pointing to the need for further investigation of its thermophysical properties in technologically relevant environments. Thus far, the eutectic composition of the binary NaCl - AmCl_3 system has been predicted to at 40-45 mol% AmCl_3 . (Toni Y Karlsson and Pinto 2024) Very limited thermophysical property data for AmCl_3 -based MSs are currently available in the literature. The lack of fundamental data will pose barriers to reactor safety assessments and fuel cycle optimization. Addressing this gap is essential for enabling the reliable deployment of chloride-based MSR technologies.

In this work, by leveraging recent advances in machine learning interatomic potentials (MLIPs), we investigated key thermophysical properties of a NaCl - PuCl_3 - AmCl_3 ternary MS with approximately 31 mol% PuCl_3 and 14 mol% AmCl_3 , including the liquid density, thermal expansion, heat capacity, viscosity, and thermal conductivity. The use of machine learning-based potentials allows for accurate, large-scale molecular dynamics simulations at a fraction of the computational cost of traditional *ab initio* methods, thereby enabling reliable predictions of temperature-dependent trends. This approach not only provides quantitative estimates of fundamental properties but also helps gain new insights into the underlying atomic-scale mechanisms governing thermal, transport and structural behavior in complex molten systems.

2.0 Methods

As demonstrated in our previous works as well as in the literature, MLIPs can significantly accelerate atomistic modeling of MS systems (Nguyen et al. 2023; Nguyen et al. 2025; Xu et al. 2023). Studies of actinide-bearing MSs stand to benefit substantially from the use of MLIPs. High-accuracy methods based on electronic structure theory, such as ab initio molecular dynamics, provide critical insight but are computationally prohibitive for the large-scale simulations required to capture transport properties complex liquids. MLIPs, by contrast, can reproduce ab initio accuracy at a fraction of the computational cost, enabling simulations that are both extensive and predictive, making them especially valuable for actinide-bearing systems, where experimental data are scarce and direct measurements are often limited.

To create a MLIP for the NaCl-PuCl₃-AmCl₃ system, the following workflow was employed. We began by conducting AIMD simulations to generate reference datasets, including atomic coordinates, energies, and forces, for the system. These datasets were then used to train a MLIP capable of reproducing the accuracy of AIMD. Finally, the trained potential was applied in large-scale molecular dynamics simulations, enabling the efficient calculation of key thermophysical properties of the mixture.

2.1 Ab initio molecular dynamics

AIMD simulations were carried out using CP2K (Kühne et al. 2020). The energy was calculated with spin-polarized revPBE-vdW (Zhang and Yang 1998; Dion et al. 2004) density functional calculations. The Gaussian and Plane-wave (GPW) hybrid basis set scheme (Lippert, Hutter, and Parrinello 1997) was employed, with the double-zeta valence polarized (DZVP) (Doudin et al. 2019; Lu et al. 2021) Gaussian basis sets and a plane wave cutoff of 600 Ry. The GTH pseudopotentials (Lu et al. 2021; Goedecker, Teter, and Hutter 1996) were used with the number of valence electrons being 1 for Na, 16 (Pu), 17 (Am) and 7 (Cl). The accuracy of this density functional theory (DFT) approach is demonstrated below for solid state systems. The initial structure and density of the system at each temperature were prepared using molecular dynamics based on the polarized ionic model (Salanne and Madden 2011). AIMD simulations were then carried out with isothermal-isobaric ensemble (NPT) in which the pressure (1 bar) and temperature were controlled with the Nose-Hoover chain barostat/thermostat (Martyna, Klein, and Tuckerman 1992). A 2-fs time step was used.

2.2 Machine learning interatomic potentials

In this work we employed the DeepMD potential model (Zeng et al. 2023) implemented in the DeePMD-kit (Zeng et al. 2023) to train MLIPs. In this approach, the energy of the system is given by

$$E = \sum_i E_i = \sum_i N(D_i(\mathbf{R}_i))$$

where E_i is the local atomic energy determined by atom i and its neighbors within a cutoff R_c , the descriptor D_i is the feature matrix encoding the surrounding environment and is fed to a deep neural network N which returns the energy E_i . $\mathbf{R}_i$ is the set coordinates of all atoms in the environment, $\mathbf{R}_i = \{\mathbf{r}_{ij} \equiv \mathbf{r}_i - \mathbf{r}_j\}$.

The network is trained by minimizing the loss function

$$\mathcal{L} = p_E |\Delta E|^2 + \frac{p_f}{3N} \sum_i |\Delta F_i|^2$$

in which ΔE and ΔF are the deviation of the potential energy and atomic forces between the reference AIMD and predicted data, respectively; and p_E and p_f are tunable pre-factors.

Here we used a radial cutoff and a smooth cutoff of 7.0 and 6.5 Å, respectively. The pre-factor p_E was set to increase from 0.02 to 1 and p_f was set to decrease from 1000 to 1. We employed 20,40,80} embedding and 250,250,250} fitting network.

From 46000 AIMD frames, 37000 frames were randomly chosen to create a training set, similarly, 4500 frames for a validation set and 4500 frames for a test set.

2.3 Molecular dynamics based on machine learning interatomic potentials

Machine learning interatomic potential molecular dynamics (MLIPs) were conducted using Large-scale Atomic/Molecular Massively Parallel Simulator (LAMMPS) (Thompson et al. 2022). To calculate the liquid density we employed the NPT ensemble in which the pressure P and temperature T were maintained with Nosé–Hoover barostat/thermostat (Evans and Holian 1985). The time step was set at 1 fs. Viscosity calculations were performed using the NVT ensemble where the temperature was controlled with the Nosé–Hoover thermostat.

2.4 Atomic systems

Here we employed cubic boxes consisting of 110 Cl, 32 Na, 18 Pu, and 8 Am atoms, Figure 1, corresponding to the composition of approximately 55 mol% NaCl, 31 mol% PuCl₃, and 14 mol%

AmCl_3 . This is equivalent to adding about 14 mol% AmCl_3 to the eutectic binary system of NaCl and PuCl_3 .

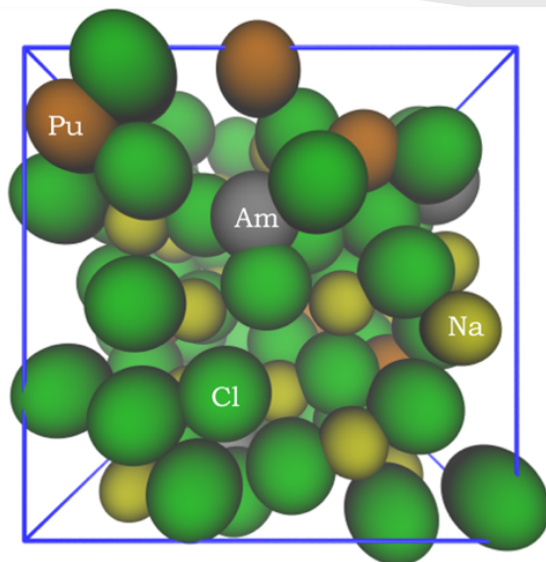


Figure 1. A simulation box with Cl in green, Na in yellow, Pu in orange, and Am in grey.

3.0 Results

3.1 Solid state systems

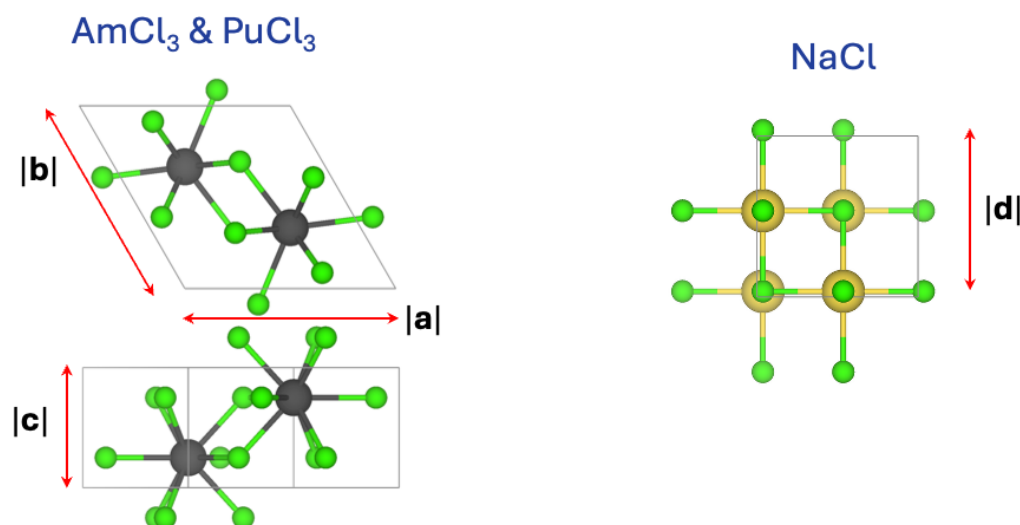


Figure 2. Solid state structures of three solids with chlorine in green and metal in grey/orange.

To test the accuracy of the DFT method in use, we first calculated the lattice constants of three solids: NaCl , PuCl_3 , and AmCl_3 . While solid AmCl_3 and PuCl_3 adopt the $P63/m$, NaCl has the $Fm-3m$ space group, Figure 2. A supercell equivalent to $2 \times 2 \times 4$ and $3 \times 3 \times 3$ primitive unit cells was used for $\text{AmCl}_3/\text{PuCl}_3$ and NaCl , respectively. Only the *Gamma* point was used to sample the Brillouin zone in DFT calculations. In the actinide chloride systems, the ferromagnetic spin configuration appeared to be slightly more stable than the antiferromagnetic one. Table 1 shows that these calculated parameters agree well with experimental data, indicating the accuracy of the DFT method employed in this work.

Table 1. Calculated and experimental lattice constants and errors. Experimental data for actinide chlorides (Asprey, Keenan, and Kruse 1965; Burns, Peterson, and Stevenson 1975) and sodium chloride (Froyen and Cohen 1986) taken the literature.

	AmCl_3			PuCl_3			NaCl		
	Cal.	Exp.	Err.(%)	Cal.	Exp.	Err.(%)	Cal.	Exp.	Err.(%)
$ a = b $ (Å)	7.337	7.390	0.7	7.330	7.394	0.8			
$ c $ (Å)	4.234	4.234	~0.0	4.296	4.234	1.3			
$ d $ (Å)							5.665	5.640	0.4

3.2 Machine learning interatomic potential training

The root mean square errors (RMSEs) for different data sets, shown in Figure 4 and Table 2, provide a quantitative measure of the accuracy of the trained interatomic potential. Compared to the AIMD reference data sets, the observed deviations are small: the energy differences are about 1 meV per atom, while the force differences are approximately 60 meV/Å. These values are well within the range typically considered acceptable, demonstrating that the trained interatomic potential can reliably reproduce AIMD-level accuracy.

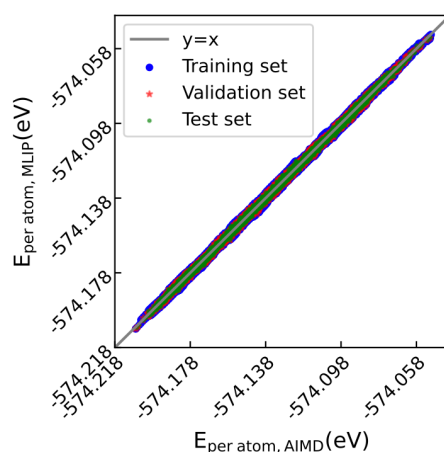


Figure 3. Potential energy calculated using the trained MLIP for different data sets versus AIMD data.

Table 2. Error of the energy (meV/atom) and force (meV/ Å) of the MLIP vs AIMD for different datasets

	Training set	Validation set	Test set
Energy RMSE	1.05	1.08	1.06
Force RMSE	59.5	59.6	59.6

3.3 Density

The density was calculated as

$$\rho = \frac{m}{V}$$

with m being the total atomic mass and V being the equilibrium volume of the simulation box. Since the mass m is independent of temperature, whereas the volume V changes with temperature, our task reduces to determining V at given temperatures. This was accomplished by using the PNT ensemble as mentioned in the method section. Figure 4 shows the variation of liquid density as a function of temperature, demonstrating the expected trend of decreasing density with rising temperature. The numerical data were fitted to a linear relationship, resulting in the equation $\rho = 4.54 - 9.4 \times 10^{-4} \times T$, $R = 0.999$.

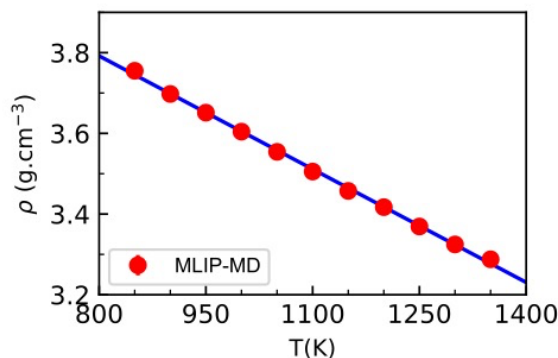


Figure 4. Temperature dependence of density.

3.4 Thermal expansion coefficient

The decrease of the density with temperature reflects the thermal expansion of the system. The thermal expansion was evaluated as

$$\beta = \frac{1}{V} \left(\frac{\partial V}{\partial T} \right)_p = -\frac{1}{\rho} \left(\frac{\partial \rho}{\partial T} \right)_p$$

which was calculated directly from the density data. Figure 5 shows that the thermal expansion is relatively small in magnitude; nevertheless, it increases progressively as the temperature

risers. This is indicative of enhanced structural flexibility at higher temperatures. Fitting the numerical data to the linear equation leads to $\beta = 1.5 \cdot 10^{-4} + 6 \times 10^{-8} T$

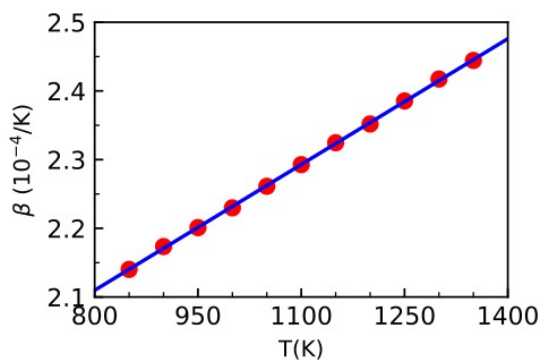


Figure 5. Thermal expansion coefficient as a function of temperature.

3.5 Constant pressure specific heat capacity C_p

We evaluated constant pressure specific heat capacity C_p by using the variation of the enthalpy with temperature:

$$C_p = \left(\frac{\partial H}{\partial T} \right)_p \approx \frac{\Delta H}{\Delta T}$$

Figure 6 shows that the enthalpy varies linearly with temperature over the examined range of temperature. It is thus straightforward to evaluate the specific heat which amounts to 0.55 J/gK which is slightly lower than measured values (~ 0.6 J/gK) of the binary NaCl-PuCl₃ with 36 mol% PuCl₃ (Toni Y. Karlsson et al. 2023), which has a lower actinide content.

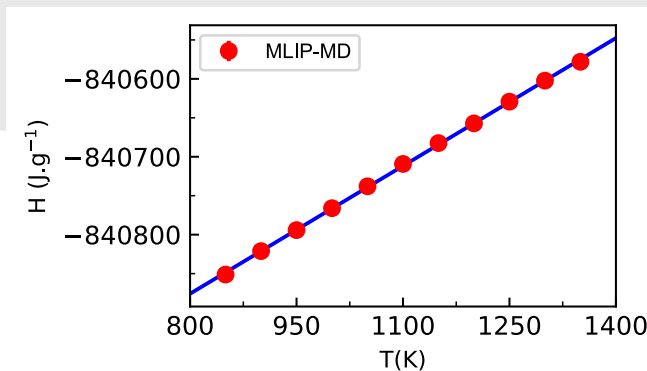


Figure 6. Temperature dependence of enthalpy.

3.6 Viscosity

We calculated the viscosity by using the Green Kubo approach

$$\eta = \frac{V}{k_B T} \int_0^t \langle P_{xy}(0) P_{xy}(t') \rangle dt'$$

in which V denotes the volume of the simulation box, k_B is the Boltzmann constant, T is the temperature, P_{xy} is an off-diagonal component of pressure tensor, and the angle brackets $\langle \dots \rangle$ indicate an ensemble average. A major challenge in this calculation is the poor convergence of η as a function of time. To mitigate this, several independent simulations were carried out at each temperature, and their results were averaged, as demonstrated in Figure 7(a).

Figure 7(b) shows the viscosity decreases rapidly with the temperature. Fitting the numerical data to an Arrhenius-type equation a functional relationship between η and T

$$\eta = A \times e^{B/k_B T}$$

we obtained 0.0044 mPa.s and 0.383 eV for parameters A and B , respectively.

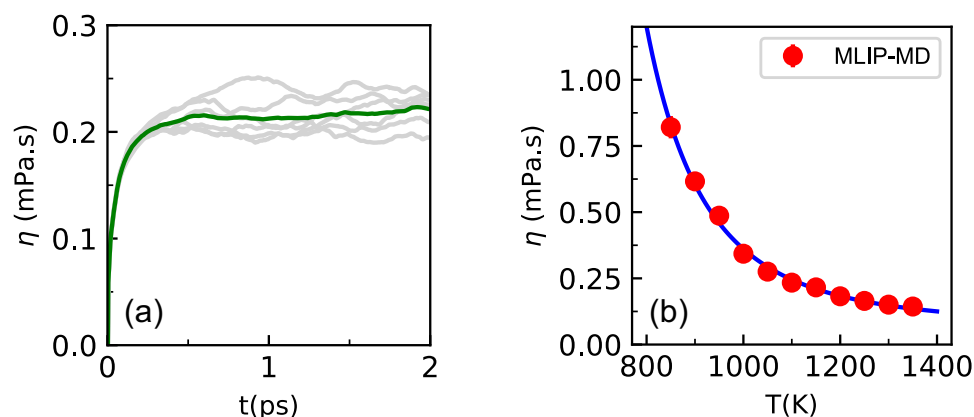


Figure 7. Viscosity as a function of time from several independent simulations (grey) and the averaged data (green) at 1150 K, (b) Temperature dependence of viscosity.

3.7 Thermal conductivity

We used the variant of Muller-Plathe reverse perturbation method (Plimpton 2014) to evaluate the thermal conductivity

$$\kappa = \frac{\Delta Q}{2 A \Delta t} \times \frac{1}{\Delta T / \Delta z}$$

in which ΔQ denotes the amount of heat added to a “hot” region and removed from a “cold” region along the z direction (Figure 8(a)), A is the cross-sectional area of the simulation box in the xy -plane, $1/\Delta t$ is the frequency at which ΔQ is added or removed; and $\Delta T/\Delta z$ represents the temperature gradient along the direction from the hot to the cold regions. In our thermal conductivity calculations, the cubic simulation box described above was extended by replicating it three times along the z -direction. Figure 8(b) shows linear changes of the temperature between the hot and cold regions which allow for evaluation of the gradient.

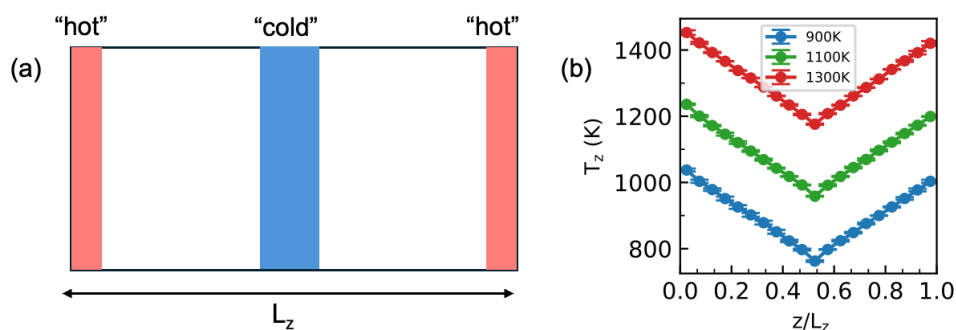


Figure 8. (a) Schematic illustration of the “hot” and “cold” regions along the z direction under periodic boundary conditions, (b) The variation of temperature along the z direction, (c) Thermal conductivity at different temperatures.

Figure 9 shows that (i) the thermal conductivity of the system is in the $0.2\text{--}0.3$ ($\text{W cm}^{-1} \text{K}^{-1}$) range and (ii) it decreases with the temperature. The thermal conductivity of this system is lower than that of pure NaCl, which has been measured to be $0.4\text{--}0.5$ ($\text{W cm}^{-1} \text{K}^{-1}$) (Harada et al. 1992). Understandably, adding more heavy atoms would reduce vibrational frequencies in the system, lowering the thermal conductivity.

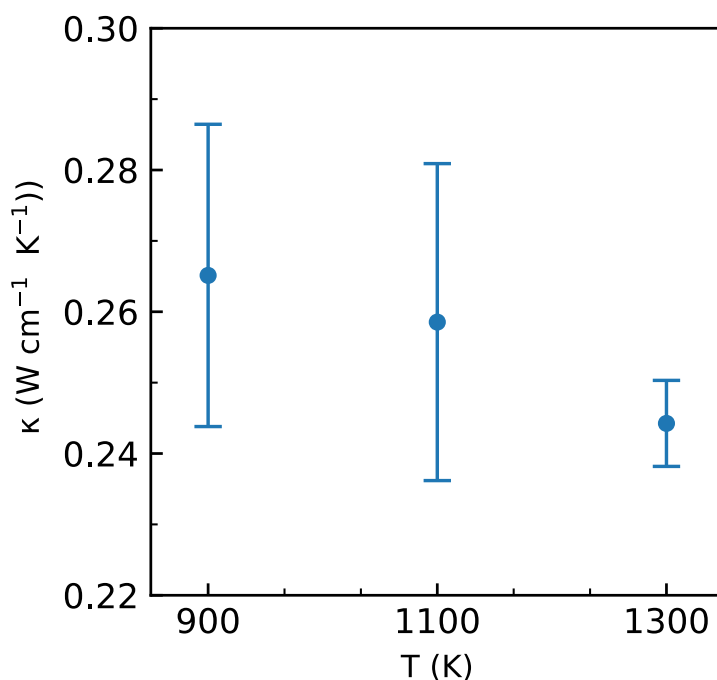


Figure 9. Thermal conductivity at different temperatures.

3.8 Structural properties

To understand how metal ions interact with the counterion Cl^- , we calculated the radial distribution functions (RDFs), $g(r)$, for cation–anion pairs in the system at 1200 K (Figure 10). The positions of the first peaks in $g(r)$ indicate that the Am–Cl and Pu–Cl bond distances are almost the same (2.73 Å), and both are slightly larger than the Na–Cl bond distance (2.71 Å). The RDFs also suggest that the Na–Cl interaction is weaker than the interactions of Am–Cl and Pu–Cl. This is in part due to the charge difference of the metal ions. The Cl^- coordination number (i.e., the ensemble-averaged number of chloride ions in the first coordination shell of each cation) is 7.0 for Pu, 7.1 for Am, and 6.2 for Na.

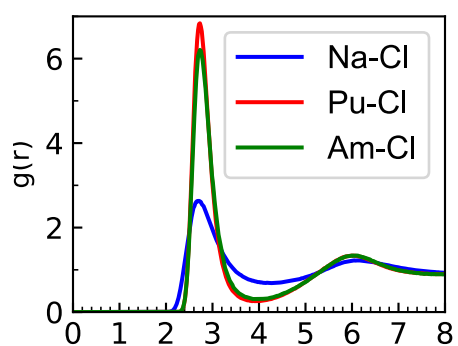


Figure 10. Radial distribution functions of cation-anion pairs.

4.0 Concluding remarks

Using molecular dynamics, we calculated the density, thermal expansion, specific heat capacity, viscosity, and thermal conductivity of a ternary system of NaCl-PuCl₃-AmCl₃ with approximately 31 mol% PuCl₃ and 14 mol% AmCl₃. Ab initio molecular dynamics was first employed to generate datasets used for training an accurate machine learning interatomic potential that allowed for extensive sampling of the system. Linear equations for the temperature dependence of the density and thermal expansion were determined. The viscosity was found to decrease exponentially with increasing temperature. Next, thermal conductivity was calculated, showing a decline with temperature. Structural properties were finally calculated to understand fundamental interactions between ions. This work provides the first evaluation of the thermophysical properties of an important molten mixture, AmCl₃-PuCl₃-NaCl, which is relevant to fast spectrum molten salt reactors. Finally, we will be able to compare calculated data in this report with experimental data being determined at Idaho National Laboratory, and this will be presented in a peer-reviewed journal article planned for FY 2026.

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