

Advanced Moderator Material Handbook

(FY22 Version: Revision 2)

**Nuclear Technology
Research and Development**

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FOREWARD TO FY22 REVISION

The first version of the “Advanced Moderator Material Handbook” was published in United States Fiscal Year 2020 with the purpose of consolidating the knowledge on yttrium dihydride for nuclear reactor moderator applications [handbook v1]. The purpose of the handbook was to create a living document to reflect updates to new findings, results, and publications of experimental data related to nuclear applications of yttrium dihydride as they become available. This revision serves as the first update based on findings that have been published since the publication of the original handbook.

The principal addition to the FY22 revision comes from the Transformational Challenge Reactor (TCR) program originally at Oak Ridge National Laboratory (ORNL). These new results are primarily in the form of material properties. Additionally, sections were split and reorganized to improve clarity and overall readability. Forthcoming updates to this handbook will include post-irradiation examination (PIE) of reactor-tested yttrium dihydride through both the DOE-NE Micro Reactor Program and the TCR program. It is intended that this handbook remain a living document with additional updates in the future.

SUMMARY

High hydrogen density moderators such as metal hydrides are an important research topic within the DOE NE Microreactor Research, Development, and Deployment (RD&D) Program due to their ability to retain hydrogen to much higher temperatures than other hydrogenous media. This class of moderators, which includes yttrium dihydride (YH_2), thermalizes neutrons in the system such that the overall fuel mass or the required uranium enrichment in the system can be significantly reduced. Knowledge of material properties, both in the as-fabricated and irradiated state, are important to understanding moderator performance during steady-state and transient reactor operation. This document provides a detailed summary of the literature data on yttrium dihydride, thermomechanical and other property data, and a critical evaluation of that data. This handbook also provides a description of ongoing experiments to understand in-reactor performance, such as irradiations in ATR, as well as nuclear data from an integral critical experiment at NCERC. This report focuses on measured quantities but also includes some modeling results for comparison where applicable.

From the evaluation of the available literature data, further work is necessary to better develop yttrium dihydride, standardize fabrication methods and experiments, and thus enable deployment as a moderator for microreactors. Although property data is relatively extensive, details on material degradation are lacking. In particular, the knowledge base for hydrogen redistribution in yttrium dihydride under temperature gradients, as well as moderator reactivity during accident scenarios such as a reactor breach, warrant further study. In addition, literature information on hydrogen concentrations at given pressures and temperatures exhibits significant variation. To that end, understanding the degradation of this material during normal and off-normal conditions is imperative to its use as a nuclear reactor moderator.

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CONTENTS

FOREWARD TO FY22 REVISION.....	iii
SUMMARY.....	iv
ACRONYMS.....	xiii
1. Introduction.....	1
1.1 Moderators for microreactor applications.....	1
1.2 Conversion constants and useful equations.....	1
1.3 Summary of measurement techniques.....	2
2. Introduction to yttrium hydride.....	4
2.1 Metal hydride moderators.....	4
2.2 Specifications.....	5
2.3 Summary of needs.....	5
2.4 Future research needs and considerations.....	6
3. Casting and fabrication techniques for yttrium dihydride.....	7
3.1 Preparation of high-purity yttrium.....	7
3.1.1 Reduction of yttrium halide with calcium.....	7
3.1.2 Yttrium-magnesium intermediate alloy process.....	7
3.2 Direct hydriding of yttrium metal.....	8
3.3 Powder metallurgy of yttrium dihydride.....	8
4. Properties of un-irradiated yttrium hydride.....	9
4.1 Phases, phase transitions, and phase diagrams.....	9
4.1.1 Phases.....	9
4.1.2 Phase diagrams.....	10
4.1.3 Structural data.....	15
4.2 Properties of un-irradiated yttrium dihydride.....	16
4.2.1 Phase formation thermodynamics.....	16
4.2.2 Phase-formation kinetics.....	20
4.2.3 Heat capacity.....	20
4.2.4 Thermal expansion and density.....	22
4.2.5 Thermal conductivity.....	24
4.2.6 Emissivity.....	25
4.2.7 Electrical properties.....	25
4.2.8 Magnetic properties.....	26
4.2.9 Microstructure.....	26
4.2.10 Mechanical properties.....	27
4.3 Hydrogen self-diffusion in yttrium hydrides.....	30
4.4 Degradation of un-irradiated yttrium dihydride.....	33
5. Neutronic considerations of yttrium dihydride.....	33
5.1 Lethargy and moderating power.....	34

5.2	Moderating ratio	37
5.3	Effect of hydrogen diffusion on neutronic properties.....	38
6.	Historical Irradiation of yttrium dihydride in FFTF.....	40
7.	Post-irradiation examination of yttrium dihydride irradiated in the ATR	41
7.1	Introduction to the irradiation experiment	41
7.2	Materials and Methods.....	44
7.3	PIE results	46
8.	Integral critical experiment at NCERC	46
8.1	Introduction to the critical experiment.....	46
8.2	Materials and methods	47
8.3	Integral critical experiment results	49
	References.....	50

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FIGURES

Figure 2.1: Hydrogen atom density as a function of temperature for water various metal hydrides in equilibrium with 1 atm of hydrogen gas for temperatures between 25 and 1400 °C (reproduced from [2]).	5
Figure 4.1: Compilation of PCT curves for the Y-H system up to a hydrogen-to-yttrium ratio of 2.0. Data reproduced from [12], [49], [50], [54], [55].	12
Figure 4.2: Redrawing of the Y-H phase diagram based on CALPHAD analysis by Peng et al. [57]. Phase regions are labeled, and phase regions have been modified from the original source as a function of H/Y atom ratio.	15
Figure 4.3: Partial molar enthalpy of formation for YH_x as a function of hydrogen content in H/Y units. Data taken from [12], [49], [54], [55], [64].	18
Figure 4.4: Partial molar non-configurational entropy of YH_x as a function of hydrogen content in H/Y units. Data taken from [12], [49], [54], [55].	19
Figure 4.5: Molar heat capacity of yttrium dihydride as a function of temperature. Data from [16]–[19], [69]. The heat capacities of end member stoichiometries from Trofimov, et al. are plotted.	21
Figure 4.6: (a) Thermal strain and (b) thermal expansion coefficient for yttrium dihydride. Thermal expansion coefficient was fit to functions of temperature, the fitting parameters for which are summarized in Table 4.4. Figure adapted from [17].	23
Figure 4.7: (a) Thermal conductivity and (b) thermal resistivity for yttrium dihydride as a function of temperature. Data from [16], [17], [19].	25
Figure 4.8: Shear, bulk, and Young's moduli as a function of hydrogen content for yttrium dihydride. Data from [2], [14], [36], [83], [85]. Young's moduli are plotted in circles, bulk moduli are plotted in triangles, and shear moduli are plotted in squares.	28
Figure 4.9: (a) Shear, (b) bulk, and (c) Young's moduli of yttrium dihydride as a function of porosity. Data from [2], [14], [36], [83], [85].	29
Figure 4.10: Magnetization evolution before and after a pulse of RF radiation is applied. The applied magnetic field is in the direction of the z-axis.	31
Figure 4.11: Loss of coherence of magnetization in the x-y plane after an RF pulse is applied.	31
Figure 5.1: Moderating power as a function of temperature for various candidate moderator materials. All data points are taken from literature for materials at 1 atm pressure [12]–[14], [47], [49], [104]–[116]. Data for metal hydrides were determined for those materials under 1 atm of pure hydrogen.	36
Figure 5.2: Hydrogen migration under the influence of temperature gradient.	39
Figure 7.1: ATR core showing LANL-MOD-1 in the B2 position.	42
Figure 7.2. Elevation stack-up of the TZM capsules in the ATR B2 position.	43
Figure 7.3. Hydride specimens located in a single ATR fixture basket. Two TZM inner capsules containing the hydride fit within a single ATR fixture, along with passive temperature and fluence monitors.	45
Figure 8.1. Critical experiment design overview.	47
Figure 8.2. C-Disc (HEU Unalloyed Metal).	48

Figure 8.3. Alumina and graphite heater to be used in integral critical experiments at NCERC.....48

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TABLES

Table 4.1: Summary of hydrogen contents for yttrium and its hydrides that are relevant to nuclear reactor moderator applications at room temperature and ambient pressure.	10
Table 4.2: Summary of equilibrium hydrogen pressures from literature PCT data [12], [49], [50], [54], [55]. Underlined values were extrapolated from low-temperature data. No data exists for $H/Y = 2.0$ due to the difficulty in achieving full stoichiometry at these temperatures.	14
Table 4.3: Lattice parameters for Y, YH_2 , and YH_3 measured experimentally at ambient temperature and pressure using X-ray (XRD) and neutron diffraction (ND). *Setoyama et al. measured the lattice parameter (a) of δ - YH_{2-x} as a function of hydrogen content that follows the empirical relation $a \text{ nm} = 0.5215 - 4.127 \times 10^{-4} cH(H/Y)$	16
Table 4.4: Empirical fitting parameters for thermal expansion coefficient as a function of temperature. Fitting parameters are given with 95% confidence intervals.	23
Table 4.5: Summary of hardness values for yttrium dihydride. Data from [2], [14], [16], [85], [86].	30
Table 4.6: Available diffusion parameters for the Y-H system. D_0 values were calculated as described in the text when not provided in the reference.	32
Table 5.1: Lethargy and average number of collisions required to thermalize 1 MeV neutrons for various nuclei of interest for nuclear reactor moderator applications.	34
Table 5.2: Moderating lethargy, powers, and ratios for various candidate moderator materials at room temperature and ambient pressure.	38
Table 5.3: Summary of the effect of hydrogen content on material and neutronic properties.	40
Table 7.1. Specimen test matrix showing the three target temperatures, fabrication methods, and characterization techniques for PIE analysis.	41
Table 7.2. Yttrium hydride physical properties.	44
Table 7.3. Total quantities of yttrium hydride for ATR PIE.	45

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ACRONYMS

ATR	Advanced Test Reactor
CTE	Coefficient of thermal expansion
DFT	Density functional theory
EFPD	Effective full power days
FFTF	Fast Flux Test Facility
HEU	High enriched uranium
HFEF	Hot Fuel Examination Facility
INL	Idaho National Laboratory
LANL	Los Alamos National Laboratory
LANSCE	Los Alamos Neutron Science Center
LEU	Low enriched uranium
LWR	Light water reactor
NASA	National Aeronautics and Space Administration
NCERC	National Criticality Experiments Research Center
NMR	Nuclear magnetic resonance
NNSS	Nevada National Security Site
PCT	Pressure-composition-temperature
PIE	Post-irradiation examination
QA	Quality assurance
SEM	Scanning electron microscopy
SNM	Special nuclear material
TEM	Transmission electron microscopy
XRD	X-ray diffraction

ADVANCED MODERATOR MATERIALS HANDBOOK

1. INTRODUCTION

1.1 Moderators for microreactor applications

Advanced nuclear reactors such as small modular (SMR) and microreactors are emerging as a viable power generation technology around the world. With their small size, high power output, and relatively low cost, microreactors are envisioned to meet localized power generation needs; these may include integration into microgrids to provide power to remote locations and disaster-prone regions, space applications such as nuclear propulsion or fission surface power, and military forward bases. The United States government as well as several private companies have expressed commitment in developing novel, compact nuclear power generation. The Kilopower Reactor Using Stirling Technology (KRUSTY) was a recent demonstration performed at the National Criticality Experiments Research Center (NCERC) in March 2018 [1]. Its ambitious, self-regulating features made it highly attractive for terrestrial and space power and propulsion applications. However, the KRUSTY design required highly enriched or weapons-grade uranium to achieve criticality, much like many existing compact designs. Reduction of the enrichment level from weapons-grade to civilian-grade fuel is highly sought after but results in higher core masses; this can be ameliorated by using a moderator to slow down neutrons. Moderator materials selected for these concepts must be useable at high temperatures to improve power conversion efficiencies, which makes metal hydrides, graphite, and beryllium-containing compounds attractive candidate materials.

Metal hydrides exhibit greater moderating efficiencies due to their high hydrogen densities. Moreover, zirconium, cerium, and yttrium hydrides retain their high hydrogen densities at temperatures in excess of 500 °C, the most stable of which is yttrium hydride up to 1350 °C [2]. Due to the historical use of graphite, beryllium, and zirconium hydride as nuclear reactor moderators, there exists a wealth of property and neutronic data for these materials. Zirconium hydride (ZrH_{2-x}) has been used for decades in reactor systems as either a separate moderator or as part of the fuel system (examples include the Russian Topaz reactors and the U.S. TRIGA reactors). However, there are knowledge gaps in the properties and performance of yttrium hydride that must be met to achieve high technical and manufacturing readiness.

This Handbook serves as a culmination of the phase stability, thermodynamic, thermophysical, mechanical, and hydrogen-diffusion properties of yttrium hydride from across published literature in the context of microreactor technology development. We note that a significant portion of these collected works originate from the US Department of Energy – Office of Nuclear Energy’s Micro Reactor Program and Transformational Challenge Reactor (TCR) program [3], [4]. This Handbook is organized as follows. First, section 1.3 contains an overview of the various experimental techniques used for property measurements. In subsequent sections, some experimental techniques and analysis methods are explained further to familiarize the reader. Property evaluations are presented in terms of the validity of the literature results. Section 2 is a collection of material properties relevant to nuclear applications. Finally, various nuclear tests of yttrium dihydride are presented, including current and planned in-reactor irradiations at various test reactors including the Advanced Test Reactor (ATR), as well as an integral critical experiment at the National Criticality Experiments Research Center (NCERC) at the Nevada National Security Site (NNSS).

1.2 Conversion constants and useful equations

Gas constant: $R = 8.314 \text{ J mol}^{-1} \text{ K}^{-1}$

Avogadro’s number: $N_A = 6.022 \times 10^{23} \text{ mol}^{-1}$

Pressure (Pa) = Pressure (torr) $\times 101325/760$

Macroscopic cross section: $\Sigma_i = \sigma_i N_A$

Thermal conductivity: $\lambda = \rho C_p D_t$

1.3 Summary of measurement techniques

This section details the various techniques used to measure the experimental data presented in the rest of this document.

Bulk yttrium hydride production. Historically, metal hydrides have been produced using a Sieverts-type apparatus, which is a gas manifold attached to a calibrated-volume reaction vessel maintained at a temperature of interest [5]. A metal sample is placed in the reaction vessel and hydrogen is titrated into the vessel in well-defined aliquots. The analysis of the subsequent gas-solid reaction is termed “Sievert’s gas absorption” from which one can determine the hydrogen content of the resultant metal hydride, the pressure-composition-temperature (PCT) relationships, and, thus, the thermodynamics of the gas-solid reaction. An alternative to the Sieverts apparatus is the powder metallurgy method, e.g., used by Shivprasad *et al.* [2]. The details of each fabrication method are given in Section 2.

Measurement of hydrogen concentration. There are several methods used to measure hydrogen concentration with varying levels of accessibility, applicability, and accuracy. Sieverts gas absorption, as discussed above, allows one to measure the number of moles of hydrogen absorbed in the sample as determined by the corresponding pressure drop in the reaction vessel; note that this assumes a correct hydrogen gas equation of state is used: systematic errors in the pressure, reaction gas manifold volume, and temperature can lead to significant uncertainty. More commonly used is the gravimetric method where the difference in mass before and after the absorption reaction is equivalent to the mass of absorbed hydrogen: here accuracy depends on that of the balance and assumes that all mass change is due to hydrogen absorption. There are also destructive methods such as inert gas fusion and vacuum hot extraction where the sample is heated to release the absorbed hydrogen, which is then quantified using various analytical techniques [6]. Among the more advanced nondestructive methods, there is neutron radiography, which measures the attenuation of a neutron beam by the hydrogenated sample and has been shown to be quantitative provided that appropriate calibrations/standards are used [7], [8]. Similarly, cold-neutron prompt gamma activation provides a unique prompt-gamma decay signature upon absorption of a neutron; the intensity of this prompt gamma line is correlated to the overall hydrogen content of the material [9], [10]. These neutron techniques, however, are inherently expensive and can be difficult to access.

Phase formation. Yttrium forms various hydride phases and H-concentrations depending on the equilibrium conditions. Measurements of the phase-formation thermodynamics include calorimetry where heat evolution/absorption is measured as a function of temperature relative to a reference temperature. The phase formation thermodynamics, determined via Sievert’s gas absorption or calorimetric methods, are used to calculate the phase diagrams. One well-established technique to do this is the Calculation of Phase Diagrams (CALPHAD) method, where thermodynamic parameters for the relevant phases are summarized in a database and a computer program is used to calculate the respective Gibb’s Free Energies [11]. The computer programs perform Gibb’s energy minimization calculations to determine the equilibrium phase diagrams given the input thermodynamic parameters. Thus, the resultant phase diagrams depend heavily on the reliability of the input thermodynamic data.

Material structure provides supplementary information for phase diagram evaluation. Structural parameters are typically determined using X-ray diffraction (XRD) or neutron diffraction (ND) [12]–[14]. In these techniques, the incident particles (X-rays or neutrons) scatter off the probed material, following Bragg’s Law, which then allows one to determine the crystal unit cell parameters through structure refinement. Moreover, neutrons are highly penetrating and hydrogen nuclei in metal hydrides exhibit very high neutron scattering cross sections compared to X rays, thus enabling bulk phase fraction measurements and occupancy factor determination.

Thermophysical properties. Thermal expansion properties, including the coefficient of thermal expansion (CTE), have been measured using dilatometric methods as well as XRD and neutron diffraction [14]–[17]. In our opinion, neutron diffraction data are preferred due to the ability to probe bulk properties as well as gather simultaneous structural and thermal expansion information. Like phase formation thermodynamics, heat capacity is also traditionally measured using calorimetric methods [16], [18], [19]. Thermal diffusivity may be measured using laser flash analysis, where the sample is exposed to a laser and the measured heat rise across the sample is an indication of the material thermal diffusivity [19], [17]. Finally, thermal conductivity may be measured either via a thermal conductivity probe or from the product of the density calculated from the CTE, the heat capacity, and the thermal diffusivity, all as a function of temperature.

Hydrogen diffusion. The magnetic properties and the hydrogen self-diffusion properties of metal hydrides, including YH_x , have been measured using nuclear magnetic resonance (NMR) [20]–[23]. NMR involves application of a magnetic field to a sample to probe the local magnetic fields about the nucleus of interest. Radio-frequency pulses are used to excite the nuclei, which relax from the excited state. The relaxation times are dependent on experimental parameters and material properties. Thus, if the experimental parameters are kept constant, the material properties, such as mobility and structure, may be determined. This technique can probe the electronic structure of various nuclei and is a bulk measurement technique since the signal is representative of the ensemble of nuclei in the system.

There are several neutron-scattering methods used to measure hydrogen diffusion rates. Quasi-elastic neutron scattering (QENS) and neutron spin-echo spectroscopy are inelastic scattering techniques by which one measures very small energy exchanges between incident neutrons and the sample's hydrogen nuclei on dynamic time scales that span hydrogen self-diffusion processes ($\sim\text{ps}-\mu\text{s}$). QENS studies have been performed for various metal hydrides to uncover ion diffusion rates and pathways [24]–[27]. Recently, in-situ and ex-situ neutron radiography has been used to quantify hydrogen concentrations and diffusion rates (via the Fick equation) in zirconium alloys [28]–[31], the principles of which could be extended to studies on yttrium hydrides [32].

Mechanical properties. The mechanical properties of metal hydrides may be determined using traditional methods like uniaxial tensile testing and creep testing. However, preparation of, e.g., dog-bone samples are very difficult due to cracking and anisotropic volume expansion of large test pieces. Acoustical measurements like resonant ultrasound spectroscopy (RUS) provide a measure of elastic properties of various metal hydrides. In the RUS technique, the sample is subject to mechanical vibrations swept through ultrasound frequencies, which then generate a series of resonances that are solutions to the equation of motion. Elastic properties are then determined by modeling the experimental spectra through either finite-element analysis or energy minimization methods.

Electrical properties. Electrical properties of metal hydrides have been measured using magnetoresistance methods, where the material is subjected to a magnetic field and the potential difference across the sample is correlated to the sample geometry and the electrical resistivity [33].

Radiation effects. The response of yttrium dihydride to in-reactor irradiations is subject to future study. The various physical properties of the unirradiated state, as detailed in this Handbook, will be compared to those in the post-irradiated condition. Samples will be irradiated under well-defined conditions to pre-specified times. Following a cooling period where the irradiated material is allowed to undergo approximately ten half-lives of the primary decay mode, the samples will again be tested for material properties. In this way, the effect of irradiation may be tied directly to physical properties.

Nuclear properties. In addition to characterization of the material properties, nuclear cross section data to predict performance of the material is important to understand. Modeling of material performance such as interatomic forces and crystal lattice parameters can occur using DFT codes [17], [2], [34]–[37]. Validation of results from DFT codes has occurred using neutron time-of-flight powder diffraction and other measurements. Of particular interest for hydride material performance in a nuclear system is the generation

of thermal scattering laws that contribute to inelastic scatter cross sections such as $S(\alpha, \beta)$ in Monte Carlo transport codes such as MCNP [38]. Once the thermal scatter laws are generated, they can be processed by codes such as NJOY to generate data for MCNP [39].

2. INTRODUCTION TO YTTRIUM HYDRIDE

2.1 Metal hydride moderators

Nuclear moderators are used to thermalize, or slow down, high-energy neutrons so that they may more efficiently participate in fission reactions in isotopes such as uranium-235, which has a high capture cross-section in the thermal-neutron energy range [45]. The nearly equal mass of hydrogen to that of a neutron enables significant energy transfer via inelastic collisions. Water is therefore a common moderator used, e.g., in light-water reactors (LWRs) due to its high hydrogen density, wide availability, and low cost. However, high-temperature water and steam pose significant risk of corrosion, and the total hydrogen density in steam and supercritical water is much lower than in liquid water and a few orders of magnitude lower than YH_{2-x} . Among the promising alternatives are metal hydrides, owing to their greater thermal stability and hydrogen mass density at high temperatures. The SMR and microreactor concepts are designed around compact nuclear cores that operate at temperatures above what is practical with water moderators. In this respect, metal hydrides are a leading candidate class of materials for moderator.

Zirconium hydride (ZrH_{2-x}) is a well-established and ubiquitous moderator material in the nuclear industry as it exhibits excellent thermal and neutronic properties at operating conditions and sustains these properties at temperatures up to ~ 650 °C. However, at higher temperatures, hydrogen will eventually desorb from ZrH_{2-x} with a corresponding decrease in moderating power. Figure 2.1 plots the equilibrium hydrogen atom density of water compared to various metal hydrides that are in equilibrium with 1 atm of hydrogen gas as a function of temperature. Here, ZrH_{2-x} shows a quadratic decrease in H density between 650 and 850 °C followed by a steep drop around 850 °C. Moreover, we note that the 1 atm of hydrogen cover gas is an optimistic scenario; therefore, it is more realistic that hydrogen dissociation will occur at lower temperatures.

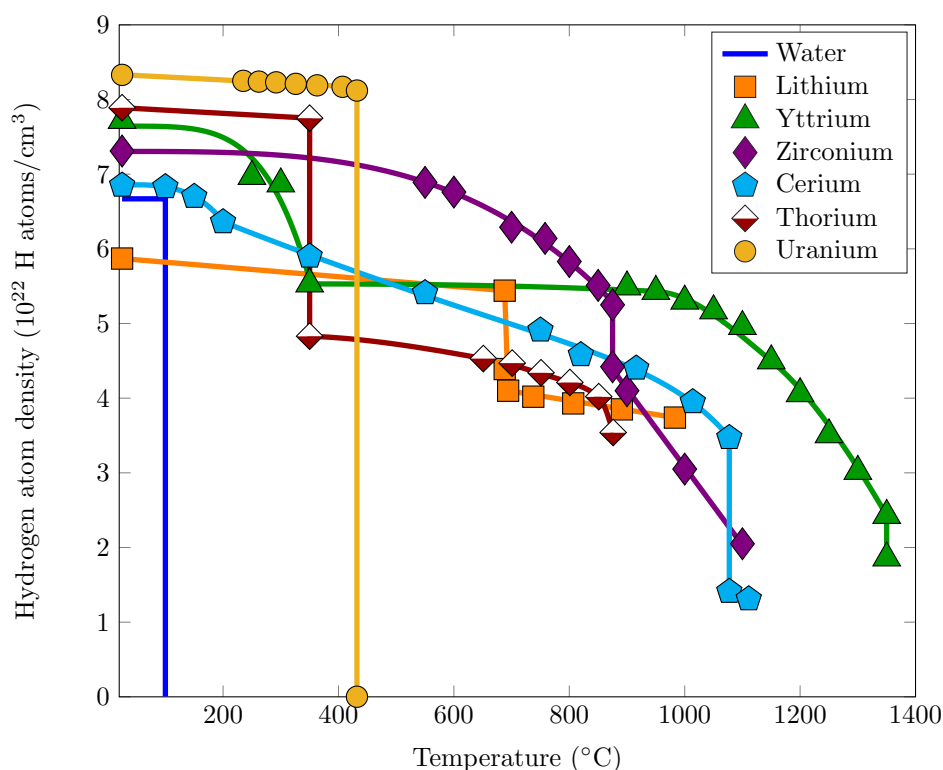


Figure 2.1: Hydrogen atom density as a function of temperature for water various metal hydrides in equilibrium with 1 atm of hydrogen gas for temperatures between 25 and 1400 °C (reproduced from [2]).

It is clear from Figure 2.1 that yttrium hydride (YH_{2-x}) displays superior hydrogen retention at the highest temperatures compared to ZrH_{2-x} . Between ~400 and 1000 °C, the H density of YH_{2-x} is approximately constant with significant H-desorption occurring around 1350 °C.

It should be noted that yttrium, which is mono-isotopic yttrium-89, has a thermal neutron absorption cross-section that is approximately an order of magnitude greater than that of natural zirconium [45]. Thus, there will be a measurable decrease in neutronic efficiency in the Y-based system. However, objectively, the effect is small. Furthermore, the ability for YH_{2-x} to perform at much higher temperatures will enable higher thermal efficiencies for power generation and, thus, reduce the mass of the power conversion and heat rejection systems [46]. The greater thermal stability of yttrium hydrides, particularly yttrium dihydride, over zirconium hydride, promotes an interest in this material for nuclear reactor moderator applications.

2.2 Specifications

The availability of high-purity yttrium stock is low given that commercial uses for pure yttrium are limited. Consequently, only a few commercial vendors have been identified that supply pure yttrium metal for the preparation of hydrides. Very-high-purity yttrium (oxygen content below 1000 wt. ppm) may only be obtained from U.S. Department of Energy laboratories such as The Ames Laboratory.

2.3 Summary of needs

The first studies of the yttrium-hydrogen system examined the thermodynamics of hydride formation with measurements of the reaction enthalpies and entropies. Likewise, X-ray and neutron diffraction studies were performed to characterize the crystal structure. The physical properties of unirradiated yttrium dihydride are well-characterized and there have been a number of studies measuring each with significant

overlap. However, phase transition kinetics are not well-characterized: that is, hydrogen absorption/desorption kinetics have not been studied for the formation of yttrium dihydride from yttrium metal. Likewise, the kinetics of the formation of yttrium trihydride have not been studied.

Broadly, the properties of unirradiated yttrium trihydride are not well-characterized. This is because synthesis of yttrium trihydride monoliths is difficult. On a per mass basis, the formation of the trihydride from yttrium metal is accompanied by a 13.7% volume increase and the formation of the trihydride from the dihydride is accompanied by an 8.9% volume increase. This results in significant structural degradation of the material and subsequent pulverization of monoliths. This will be discussed in the context of reactor design and operation.

Yttrium dihydride has been used as a moderator in previous reactors, as will be discussed in Section 6. However, as of publication of this Handbook, Post-Irradiation Examination (PIE) has not been reported on yttrium hydrides. As such, irradiation effects on microstructure, thermophysical and mechanical properties, dimensional instability, and any potential moderator-cladding interactions have yet to be evaluated. This appears to be the largest gap in our current understanding of material performance. However, irradiations of yttrium dihydride are anticipated to fill these gaps.

An integral critical experiment has been planned at the National Criticality Experiments Research Center (NCERC) at the Nevada National Security Site (NNSS) to study neutron flux as a function of temperature.

2.4 Future research needs and considerations

The deployment of yttrium dihydride as a nuclear reactor moderator is contingent upon its in-reactor performance. That is, it is necessary to understand the mode(s) of degradation that may occur under normal, transient, and accident reactor conditions. To that end, PIE of neutron-irradiated yttrium dihydride will provide insight into the failure mechanisms as well as guide mitigation strategies.

In situ measurements of hydrogen dynamics under reactor-relevant conditions are important for understanding reactivity changes during reactor operation. Hydrogen dynamics can include hydrogen redistribution within a single hydride component due to thermal gradients, but also hydrogen desorption from the hydride. Both can be predicted using modeling tools [Swift]. However, the lack of a consensus for the PCT curves in the Y-H system (as shown below) and an incomplete understanding of surface reaction rates and hydrogen self-diffusion in the hydride make modeling efforts challenging. As such, detailed studies of these parameters are imperative for the qualification of yttrium dihydride as a moderator. This will include a standardization of Sieverts-type measurements of the Y-H system, including surface preparation, microstructure, material and gas purity, pressure and temperature control, and manifold volume calibration.

Finally, material purity and microstructure are important to assess, as well as their impacts on physical and chemical properties. The difficulty in producing high-purity yttrium also necessitates research to improve production scale at facilities such as Ames Laboratory.

3. CASTING AND FABRICATION TECHNIQUES FOR YTTRIUM DIHYDRIDE

3.1 Preparation of high-purity yttrium

The first preparation of yttrium metal was by the reduction of yttrium chloride by potassium metal in 1828 by Wohler [40]. Since then, all subsequent methods of preparing yttrium metal have been variations of the metallothermic reduction of yttrium halides. The Ames Laboratory led the production of yttrium metal based on demand from the US Atomic Energy Commission as part of the Air Force Nuclear Propulsion Program. Researchers at Ames Lab developed several methods of preparing yttrium metal from yttrium halides. Detailed descriptions of these methods may be found elsewhere [41], but some of these methods are briefly summarized here.

The starting material for yttrium metal production is yttrium(III) oxide. High-purity rare-earth (including yttrium) oxides are prepared using ion-exchange methods. The oxide is then converted to a rare-earth halide, such as yttrium fluoride (YF_3) or yttrium chloride (YCl_3). High-purity yttrium fluoride is prepared by Ames Lab by direct reaction of yttrium(III) oxide with high-purity anhydrous hydrogen fluoride (HF) gas. Commercially, yttrium fluoride may be prepared by the reaction of yttrium(III) oxide with ammonium bifluoride ($\text{NH}_4\text{F}\cdot\text{HF}$). Yttrium metal resulting from yttrium fluoride produced via ammonium bifluoride has been shown to have a higher oxygen content, as compared with material produced via the gaseous hydrogen fluoride route [41]. Yttrium chloride may be prepared by the reaction between yttrium(III) oxide and either chlorine gas (Cl_2) or carbon tetrachloride (CCl_4).

The high-purity yttrium halides are further purified. Yttrium fluoride may be purified using a molten-salt method, where the fluoride is mixed with a fluxing salt and then exposed to anhydrous hydrogen fluoride gas. Yttrium chloride, on the other hand, may be vacuum distilled or may be melted and then filtered under an argon atmosphere.

3.1.1 Reduction of yttrium halide with calcium

The primary method of producing high-purity yttrium metal is reduction of yttrium fluoride or yttrium chloride using calcium or lithium metal [41]. In this method, a charge of compacted yttrium fluoride and calcium is placed in a tantalum crucible and heated to approximately 1600 °C. All processing is done in an inert environment, due to the ability of calcium to oxidize rapidly. At the Ames Lab, calcium metal is distilled under helium and stored in a glovebox for further processing. Tantalum crucibles generally adhere to ASTM B708 – 12 (current standard) [42, p. 10].

After the charge is cooled, the slag may be removed, and the yttrium remelted under vacuum and cast as an ingot.

3.1.2 Yttrium-magnesium intermediate alloy process

Another method of producing high-purity yttrium metal is through the formation of an yttrium-magnesium alloy. In this process, a charge of high-purity yttrium fluoride, calcium chloride (CaCl_2), calcium metal, and magnesium metal, is heated in a zirconium crucible at 1000 °C. At this temperature, an yttrium-magnesium alloy forms and separates from the slag. Heating to 1200 °C enabled further separation of magnesium and unreacted calcium. Heating cycles were employed to further separate the magnesium and calcium. The use of a calcium-lithium alloy as a co-reductant has also shown promising results due to the formation of a low melt-point mixture of calcium fluoride and lithium fluoride. The resultant yttrium metal may be cast or extruded into desired shapes.

3.2 Direct hydriding of yttrium metal

Direct, or massive, hydriding of high-purity yttrium metal is well-documented in the literature. Mueller et al., showed high-quality, directly-hydrided yttrium metal produced as part of the Air Force Nuclear Propulsion Program [5, p. 13], [43, p. 14]. More recently, Setoyama et al., Ito et al., Shivprasad et al., and Hu et al. have shown the ability to make high-quality yttrium dihydride by the direct hydride method [2], [14], [19], [44].

In the method detailed by Hu et al., crack-free yttrium dihydride was fabricated by exposing high-purity yttrium to ultra-high purity hydrogen gas at low flow rates and tuning the hydrogen partial pressure to the temperature to maintain a PCT relationship as per the PCT diagrams. As a result, the material is cooled on the relevant isochore. The full details of the hydriding procedure may be found in [44].

3.3 Powder metallurgy of yttrium dihydride

Shivprasad et al. have shown that powder metallurgy processes to produce yttrium dihydride are possible [2]. Mechanical and thermophysical property measurements of the sintered monoliths showed results consistent with those obtained from directly-hydrided yttrium, indicating the feasibility of the powder metallurgy process to produce high-quality yttrium dihydride. Further details of these methods may be found in [2].

Powder metallurgical methods for the production of yttrium dihydride have several advantages:

- Yttrium castings tend to have large, directionally solidified grains. Direct hydriding leads to a phase change accompanied by a large, anisotropic volume expansion. This may result in cracks and large residual stresses in the final component.
- The anisotropic volume expansion results in final parts with unpredictable sizes, which necessitates post-hydriding machining that has the potential for safety incidents due the pyrophoricity of yttrium dihydride powders.
- Processing of castings is required to refine the grain size and improve homogeneity.
- Powder processing of yttrium dihydride results in final products that are independent of pre-existing yttrium metal microstructure and physical form. Therefore, yttrium dihydride of irregular sizes and geometries may be used in lieu of large castings.
- Fabrication of composite microstructures is relatively straightforward.

4. PROPERTIES OF UN-IRRADIATED YTTRIUM HYDRIDE

4.1 Phases, phase transitions, and phase diagrams

Reviews of the Y-H system have been performed in the past, such as by Mueller et al. and Khatamian et al. [13], [43, p. 10]. In this section, we combine the data given in these reviews with other recently reported results to provide a complete picture of the current understanding of the yttrium-hydride system.

4.1.1 Phases

4.1.1.1 Stable phases

For nuclear reactor moderator applications, the relevant phases within the Y-H system can be down-selected to those stable at ambient pressure and temperatures between 25 and 1000 °C. There are three principal phases of interest: hcp yttrium metal (α -Y), fcc yttrium dihydride (δ -YH₂), and trigonal yttrium trihydride (ϵ -YH₃) [13], [43], [47]. There also exists a high-pressure yttrium trihydride, which maintains the same crystal structure as the ambient-pressure dihydride [48].

α -Yttrium metal is stable at ambient pressure up to approximately 1480 °C, at which point it transforms to the bcc β -yttrium. The melting point of yttrium metal is approximately 1530 °C. The hydrides of yttrium are formed upon exposure of α -yttrium to hydrogen gas at pressures up to 1 atm and temperatures up to approximately 1350 °C [12], [49], [50]. Hydride formation may be possible at higher temperatures. However, investigations of hydriding have not been performed to temperatures above 1350 °C.

4.1.1.2 Other phases

High-pressure compression (>17 GPa) of YH₃ or Y-metal in H₂ at elevated temperature yields hydrides with much higher hydrogen content, e.g., bct-YH₄, bcc-YH₆, and hcp-YH₉ [51], [52]. However, these “super-hydride” phases have not been shown to be stable at ambient pressures or temperatures.

4.1.1.3 Reporting H composition

The hydrogen content of metal hydrides for moderator applications has been denoted by four principal units: (1) the hydrogen-to-metal atom ratio, or H/M; (2) the hydrogen atom fraction/percent; (3) the hydrogen weight fraction/percent; and (4) the hydrogen atom density in atom/cm³. For neutronics calculations, the hydrogen atom density is the preferred unit, as it can readily be used to calculate macroscopic cross-sections.

The hydrogen-to-metal atom ratio is calculated as the moles/atoms of hydrogen absorbed by the hydride to the moles/atoms of yttrium metal in the original sample. Both forms of hydrogen-to-metal atom ratios are typically calculated through mass-balance measurements, but the moles of hydrogen absorbed can also be measured by hydrogen pressure drops during Sievert’s gas absorption.

The hydrogen atom fraction or percent is calculated quite simply from the H/M ratio, x . In one molecule of YH _{x} , the hydrogen atom fraction is given as:

$$c_H = \frac{x}{1 + x} \quad (1)$$

Hydrogen weight fraction, or percent, are similarly determined. If x is the H/M ratio, then the hydrogen weight fraction, w_H , is calculated as:

$$w_H = \frac{M_H x}{M_Y + M_H x} \quad (2)$$

where M_H is the molar mass of hydrogen (approximately 1.01 g/mol) and M_Y is the molar mass of yttrium (approximately 88.91 g/mol).

The hydrogen atom density, N_H , is calculated from the mass density of the material (either the metal with hydrogen in solid-solution or the metal hydride), ρ , and the H/M ratio, x , as follows:

$$N_H \left(\frac{\text{atoms H}}{\text{cm}^3} \right) = \frac{\rho \left(\frac{\text{g}}{\text{cm}^3} \right) x N_A}{88.91 + 1.01x} \quad (3)$$

where N_A is Avogadro's number. Table 4.1 shows a comparison between several different values for these parameters for yttrium and its hydrides at room temperature.

Table 4.1: Summary of hydrogen contents for yttrium and its hydrides that are relevant to nuclear reactor moderator applications at room temperature and ambient pressure.

x (H/M)	c_H (atom %)	w_H (weight %)	N_H (10^{22} atoms H/cm ³)	Phase(s) (-)
0.00	0.00%	0.00	0.00	α -Y
0.10	9.09%	0.11	0.30	α -Y
0.20	16.67%	0.23	0.61	α -Y
0.30	23.08%	0.34	0.91	α -Y + δ -YH ₂
0.50	33.33%	0.56	1.51	α -Y + δ -YH ₂
1.00	50.00%	1.12	3.00	α -Y + δ -YH ₂
1.50	60.00%	1.68	4.48	α -Y + δ -YH ₂
1.70	62.96%	1.89	5.06	α -Y + δ -YH ₂
1.80	64.29%	2.00	5.35	α -Y + δ -YH ₂
1.90	65.52%	2.11	5.64	α -Y + δ -YH ₂
2.00	66.67%	2.22	5.93	δ -YH ₂
2.50	71.43%	2.76	7.38	δ -YH ₂ + ϵ -YH ₃
3.00	75.00%	3.30	8.80	ϵ -YH ₃

4.1.2 Phase diagrams

The equilibrium phases of the Y-H system are summarized in a phase diagram that is generated by measuring the pressure-composition-temperature (PCT) isotherms. The equilibrium partial pressures of hydrogen are what is required to hydride yttrium to a particular stoichiometry (H/Y) at a given temperature; both are detailed in the subsequent sections.

4.1.2.1 Pressure-composition-temperature diagrams

PCT curves plot equilibrium partial pressure as a function of composition/stoichiometry for gas-solid reactions and are used extensively for hydrogen-solid interactions. These diagrams are typically developed using a Sievert's apparatus or a modification thereof. In a typical experiment, a sample is placed in a reaction vessel with a well-defined volume that is attached to a gas manifold also with a well-defined volume. The reaction vessel is evacuated to a suitable vacuum level and then heated to a desired temperature. Hydrogen is then aliquoted at precisely controlled pressures, which is followed by absorption of H₂ by the sample and the corresponding pressure drop over time is measured. The moles of hydrogen absorbed by the sample are calculated from the measured pressure drop, the system volume, and the known temperature using the Ideal Gas Law with assumptions therein. The moles of hydrogen absorbed are related

to the initial moles of sample to produce a hydrogen composition. In this way, the equilibrium pressure of hydrogen is plotted as a function of hydrogen composition for each isothermal gas absorption experiment.

It is important to note that chemical compatibility between the sample and the reaction vessel must be considered, and potential hydrogen permeation through the reaction vessel must be addressed at high temperatures. Additionally, the condition of the sample surface such as surface-adsorbed species, chemical treatment of the surface, ball-milling, and the presence of a surface oxide have been shown to impact hydriding [53].

A compilation of the PCT curves in the literature for the Y-H system is plotted in Figure 4.1 as equilibrium pressure of hydrogen (on a logarithmic scale) as a function of hydrogen-to-yttrium atom ratio. Each curve represents data collected at a particular temperature. It is observed that, for low hydrogen contents, up to $H/Y = 0.3-0.4$, the PCT data follow an approximate Sievert's law dependence, which is characteristic of hydrogen solubility in the base yttrium metal, as will be discussed in Section 4.2.1.1. For hydrogen contents between approximately $0.3 - 0.6$ and $1.0 - 1.6$ H/Y units, depending on the temperature, the hydrogen partial pressure remains constant as a function of hydrogen content, which is due to the formation of the dihydride from hydrogen-saturated yttrium metal and hydrogen gas. This regime is called the 'plateau region' and represents two-phase equilibrium between the hydrogen-saturated metal and the dihydride, following the phase rule. Finally, at higher hydrogen contents, the hydrogen partial pressure rapidly increases as a function of hydrogen content. At this point, the system is single-phase dihydride that is nearly stoichiometric, and the accommodation of further hydrogen becomes increasingly difficult, i.e., more thermodynamically unfavorable.

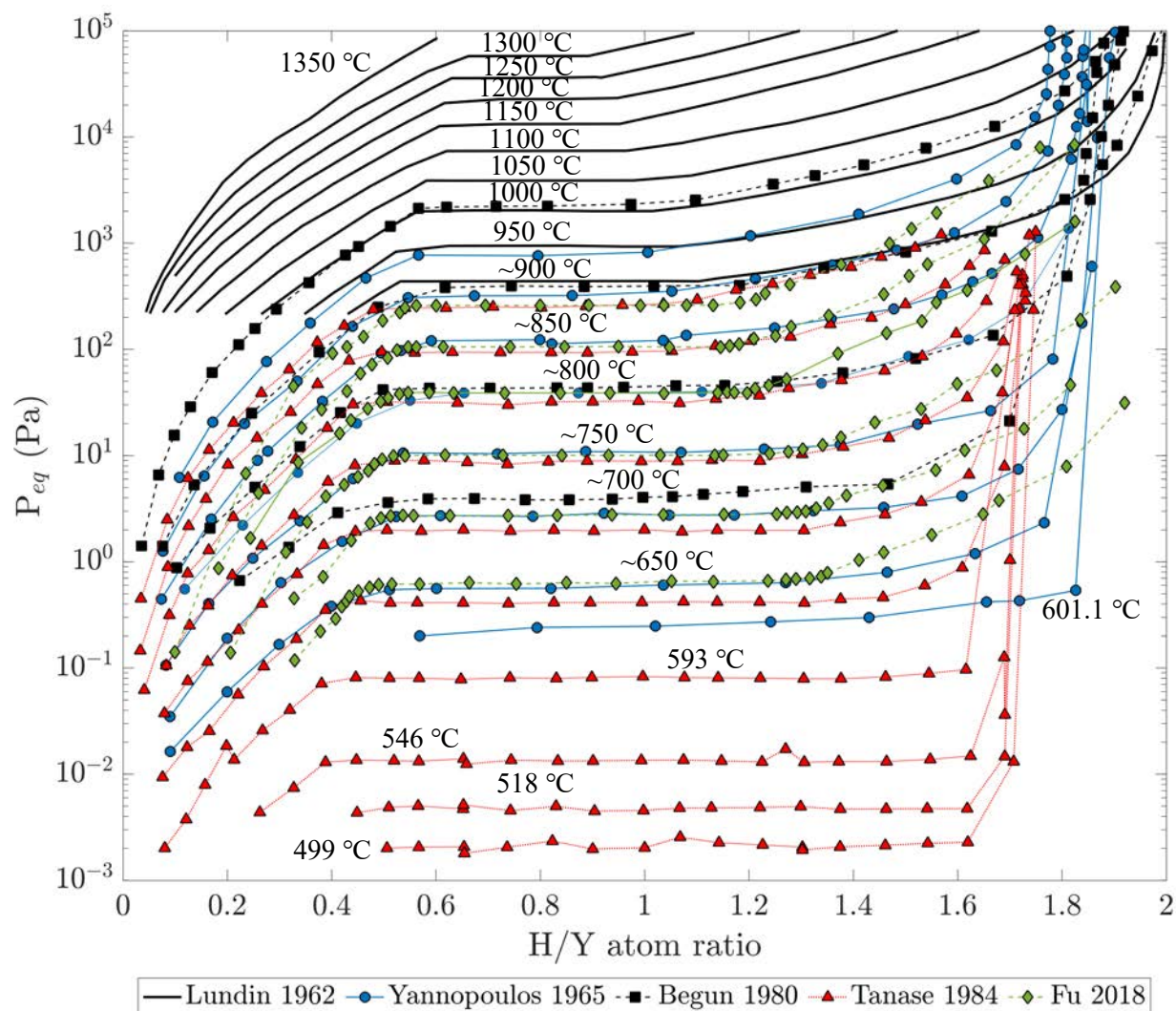


Figure 4.1: Compilation of PCT curves for the Y-H system up to a hydrogen-to-yttrium ratio of 2.0. Data reproduced from [12], [49], [50], [54], [55].

Across the literature, it is observed that the plateau regions are very consistent and as such the reproducibility is high. However, the single-phase regions (hydrogen dissolved in yttrium metal or the single-phase dihydride) are quite inconsistent. For example, in the studies by Lundin et al. and Begun et al., the terminal hydrogen content is $H/Y = 2.0$, while for Yannopoulos et al. and Tanase et al., their terminal values were closer to $H/Y = 1.90$ and 1.70 , respectively. In the single-phase region with low hydrogen contents, the relationships between the hydrogen partial pressure and hydrogen-to-yttrium atom ratio are also not consistent across the various studies, though they are consistent within each single study. The origin of the inconsistencies is not currently known; however, we speculate that the starting metal's purity is a likely cause. Lundin et al. noted that the principal impurities in their yttrium metal were 0.57% zirconium and 0.32% oxygen. Yannopoulos et al., on the other hand, noted low-Z impurities of 0.179% oxygen and 0.0033% nitrogen, by mass. The yttrium metal used by Begun et al. showed approximately 0.1% dysprosium, cerium, and zirconium, as well as 0.45% oxygen and 0.05% nitrogen, hydrogen, and other rare earths, by mass, with impurities totaling approximately 0.85% by mass. Tanase et al. measured oxygen and nitrogen impurities to be 0.083% and 0.12%, by mass, respectively. Fu et al. noted that the metal used in their work was vacuum distilled and provided a maximum purity (lowest impurity content of the group) of

0.0155% oxygen and 0.0030% nitrogen, by mass. Small differences in gas purities could affect these impurity content values. Lundin et al. obtained hydrogen source gas by thermal decomposition of uranium hydride, whereas the studies by Yannopoulos et al. and Begun et al. used high-purity hydrogen gas cylinders with impurities of approximately 60 and 10 ppm, respectively. Yannopoulos et al. further purified their gas by using a liquid-nitrogen trap and by passing the gas over zirconium turnings. Tanase et al. used gas with a purity of 99.9999%, while Fu et al. did not mention the purity of the hydrogen gas used to develop the PCT curves.

Another possible source of the discrepancies could arise from the differences in the Sievert's apparatuses used in the various studies. Lundin et al., Yannopoulos et al., Begun et al., and Tanase et al. all used ceramic reaction vessels, such as mullite and quartz, due to the low permeability of hydrogen through these materials [56]. However, Fu et al. used a stainless-steel reaction vessel, which is known to permeate hydrogen at elevated temperatures [56], raising the question as to how pressure equilibrium at high temperature was established in their study. As a result of the inconsistencies across datasets and the potential impact of gas and metal starting purities, it is believed that more work is necessary to further characterize the PCT relationships in the single-phase regions with careful consideration of sample preparation, vessel calibration, source-gas purity, and pressure/temperature control and sensitivity.

4.1.2.2 Hydrogen partial pressures

As mentioned in Section 4.1.2.1, the PCT data for the Y-H system exhibit consistency in the two-phase region (metal and dihydride in equilibrium) but show significant differences in the single-phase regions. A summary of the equilibrium partial pressures, along with the associated data spread from averaging over the various literature sources is shown in Table 4.2. Data are presented as equilibrium hydrogen pressure (in Pa) as a function of hydrogen-to-yttrium atom ratio and temperature (in °C).

Table 4.2: Summary of equilibrium hydrogen pressures from literature PCT data [12], [49], [50], [54], [55]. Underlined values were extrapolated from low-temperature data. No data exists for H/Y = 2.0 due to the difficulty in achieving full stoichiometry at these temperatures.

X (H/Y) / T (°C) / P _{eq} (Pa)	600	800	1000	1200
0.2	1.80×10^{-2}	2.43 ± 1.44	8.91×10^1	1.12×10^3
0.4	7.43×10^{-2}	$1.80 \times 10^1 \pm 7.89$	$6.48 \times 10^2 \pm 1.97 \times 10^1$	5.54×10^3
0.6	$1.43 \times 10^{-1} \pm 1.75 \times 10^{-1}$	$3.74 \times 10^1 \pm 9.59$	$2.09 \times 10^3 \pm 2.24 \times 10^2$	1.93×10^4
0.8	$1.61 \times 10^{-1} \pm 2.23 \times 10^{-1}$	$3.82 \times 10^1 \pm 9.45$	$2.14 \times 10^3 \pm 2.76 \times 10^2$	2.27×10^4
1.0	$1.65 \times 10^{-1} \pm 2.27 \times 10^{-1}$	$3.89 \times 10^1 \pm 9.78$	$2.19 \times 10^3 \pm 5.12 \times 10^2$	2.62×10^4
1.2	$1.74 \times 10^{-1} \pm 2.59 \times 10^{-1}$	$4.17 \times 10^1 \pm 9.64$	$2.98 \times 10^3 \pm 7.83 \times 10^2$	4.31×10^4
1.4	$1.87 \times 10^{-1} \pm 2.97 \times 10^{-1}$	$7.19 \times 10^1 \pm 4.72 \times 10^1$	$4.71 \times 10^3 \pm 1.37 \times 10^3$	7.61×10^4
1.6	$2.42 \times 10^{-1} \pm 4.08 \times 10^{-1}$	$1.78 \times 10^2 \pm 2.07 \times 10^2$	$8.70 \times 10^3 \pm 3.71 \times 10^3$	<u>1.40×10^5</u>
1.8	5.11×10^{-1}	$1.02 \times 10^3 \pm 9.63 \times 10^2$	$2.23 \times 10^4 \pm 1.18 \times 10^4$	<u>2.44×10^5</u>
2.0	N/A	N/A	N/A	N/A

4.1.2.3 Phase diagrams

The phase diagrams for gas-solid systems are generated by projecting PCT curves onto the T - x axis and outlining phase-stability regions. As such, the phase diagrams omit the pressures required, but represent useful information. Phase diagram analysis of the Y-H system was first presented by Yannopoulos et al. and then summarized by Mueller by compiling data from Lundin et al. and Yannopoulos et al. [43], [49]. More recently, CALPHAD analysis of the Y-H system thermodynamics was carried out by Fu et al. and Peng et al. [50], [57]. A redrawing of the Y-H phase diagram based on the analysis by Peng et al. is presented in Figure 4.2. The phase diagram from this study was used because it is consistent with the previous analyses of the Y-H phase system by Mueller and by Fu et al. It should be noted that very little data exists for yttrium dihydride at temperatures below approximately 500 °C, especially at/near the phase boundary between the two-phase and single-phase (dihydride) regions; this region is subject to further study.

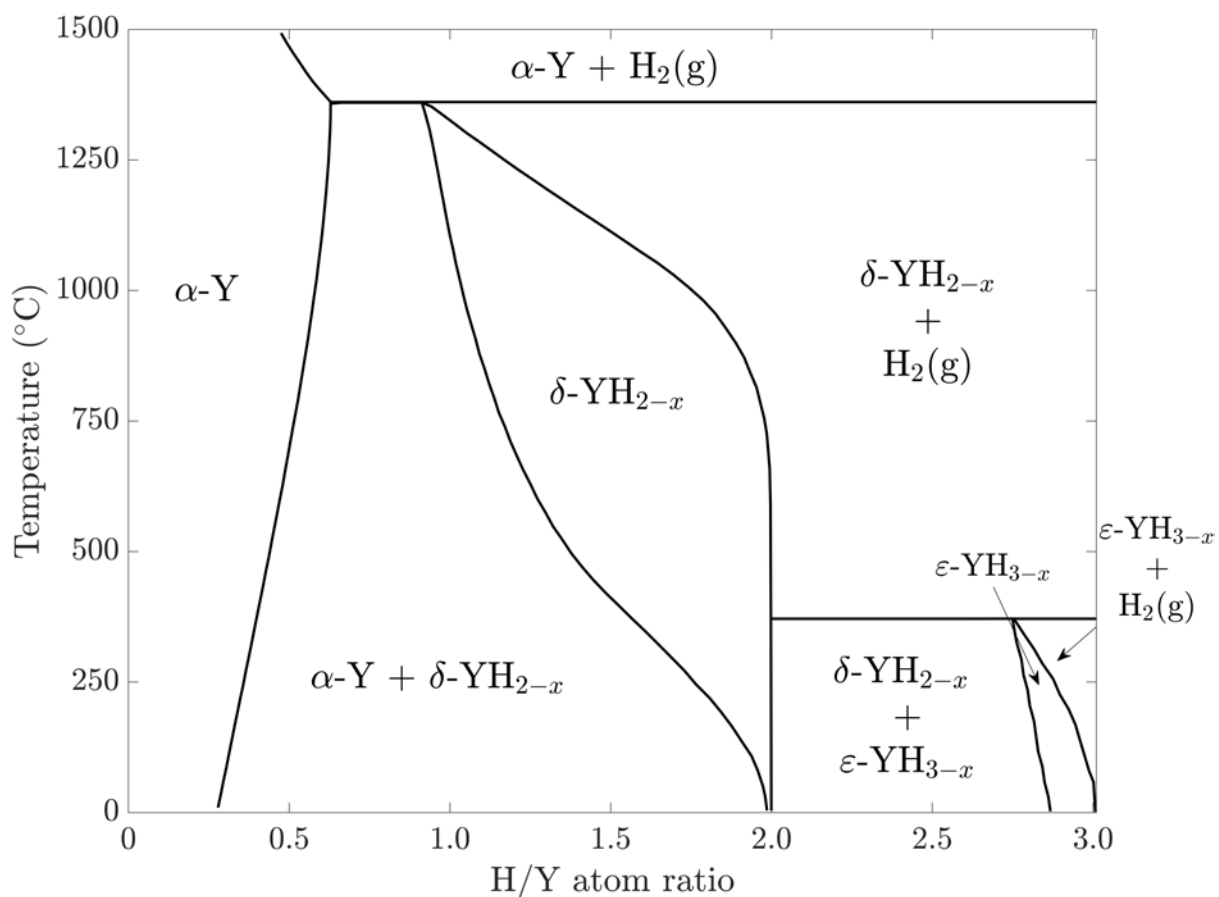


Figure 4.2: Redrawing of the Y-H phase diagram based on CALPHAD analysis by Peng et al. [57]. Phase regions are labeled, and phase regions have been modified from the original source as a function of H/Y atom ratio.

4.1.3 Structural data

Key structural parameters for the relevant stable phases of yttrium hydride are given in Table 4.3 along with the hydrogen-to-yttrium atom ratio, measured lattice parameters, and the techniques used to determine the lattice parameters. The data summarized in Table 4.3 show that the lattice parameters of each phase are very consistent across all studies, with discrepancies only principally arising on the order of approximately one-hundredth or one-thousandth of an angstrom. However, these differences could also mainly be due to differences in hydrogen content. While a more detailed study may be of interest, the data in the literature are thought to be sufficient due to the consistency across multiple studies and techniques.

Table 4.3: Lattice parameters for Y, YH₂, and YH₃ measured experimentally at ambient temperature and pressure using X-ray (XRD) and neutron diffraction (ND). *Setoyama et al. measured the lattice parameter (a) of δ -YH_{2-x} as a function of hydrogen content that follows the empirical relation a (nm) = 0.5215 – 4.127 × 10⁻⁴ c_H (H/Y).

Compound	H/Y	a (Å)	c (Å)	Technique.	Ref.
α -Y	0	3.6474 ± 0.0007	5.7306 ± 0.0008	XRD	[58]
	0	3.654	5.7501	XRD	[12]
	0.23	3.6636 ± 0.0009	5.7900 ± 0.0013	XRD	[59]
δ -YH ₂	2.00	5.201	-	XRD	[12]
	1.96	5.205 ± 0.002	-	ND	[13]
	1.7 – 2.0	*	-		[14]
ϵ -YH ₃	3.00	3.674	6.599	XRD	[12]
	2.90 – 3.00	3.672	6.659	XRD	[60]
	N/A	3.67 ± 0.02	6.62 ± 0.02	XRD	[61]

4.2 Properties of un-irradiated yttrium dihydride

4.2.1 Phase formation thermodynamics

An analysis of the PCT curves allows for calculation of the phase formation thermodynamics for yttrium or yttrium hydrides at a particular stoichiometry. The PCT curves in Figure 2.2 can be broken down into three distinct regions that span H/M ratios of 0 to 2.0: hydrogen dissolved in yttrium metal, two-phase mixture of α -Y and δ -YH₂, and single-phase δ -YH₂.

4.2.1.1 Hydrogen in yttrium metal

For low values of H/Y, hydrogen absorbs into the yttrium metal by the process of adsorption, dissociation, and dissolution (solvation of hydrogen) [62]. The chemical equation for this reaction is given as:



where H_M is hydrogen adsorbed onto the yttrium metal surface. At equilibrium, the chemical potential of the gas is equal to the chemical potential of the hydrogen in the metal. The equilibrium constant, called the Sievert's constant, is given as follows:

$$K_s = c_H \sqrt{p_{H_2}} \quad (5)$$

where K_s is the Sievert's Law constant, c_H is the concentration of hydrogen in solution, and p_{H_2} is the equilibrium partial pressure of hydrogen. K_s is related to thermodynamic properties through the Gibb's Free Energy as follows:

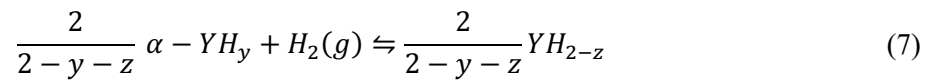
$$K_s = K_s^0 \exp\left(-\frac{\Delta H_s}{RT} + \frac{\Delta S_s}{R}\right) \quad (6)$$

where K_s^0 is the pre-exponential factor for the Sievert's constant and ΔH_s and ΔS_s are the enthalpy and non-configurational entropy of solution, respectively. Combining equations (5) and (6) enables calculation of the standard enthalpy and entropy of formation for a given H/M value given knowledge of K_s^0 [63]. The result is that Gibb's Free Energy, and thus the enthalpy and non-configurational entropy of solution, is a linear function of the hydrogen concentration.

4.2.1.2 Two-phase (α and δ) region

At intermediate values of H/Y, the hydrogen reacts with saturated yttrium metal to form the dihydride. This is represented in the PCT curves as the plateau region and consists of two-phase saturated α -Y and δ -YH_{2-z} at the equilibrium stoichiometry given by the composition at the phase boundary. For example, at 600 °C, at an overall stoichiometry of 1.0 H/Y units, the material will consist of α -Y with an approximate composition of 0.45 H/Y units and δ -YH_{2-z} with a composition of 1.5 H/Y units.

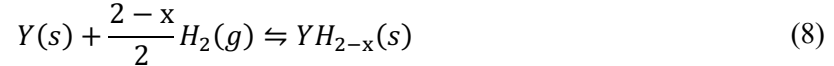
In this two-phase region, the hydrogen absorption reaction can be written as:



where y is the maximum solubility of hydrogen in yttrium metal and z is the sub-stoichiometry in hydrogen sublattice of yttrium dihydride [64]. The enthalpy and entropy of the reaction do not vary with hydrogen concentration, as required by chemical equilibrium [65].

4.2.1.3 Single-phase yttrium dihydride

The general reaction between yttrium metal and hydrogen gas to form the single-phase dihydride is given as follows:



For a reversible reaction, the equilibrium constant for this equation is given as follows:

$$K_{eq} = \frac{a(YH_{2-x})}{a(Y) a(H_2)^{\frac{2-x}{2}}} \quad (9)$$

where $a(i)$ is the activity of species i . Assuming that the activities of the solid, pure phases are unity, and that the concentrations of gases are equal to their equilibrium partial pressures ($p_g = p_{eq}/p_0$), then

$$K_{eq} = p_{H_2}^{-\left(\frac{2-x}{2}\right)} \quad (10)$$

Combining the definitions of the Gibb's free energy yields:

$$\ln(K_{eq}) = -\frac{\Delta H^\ominus}{RT} + \frac{\Delta S^\ominus}{R} \quad (11)$$

$$\frac{2-x}{2} \ln(p_{H_2}) = \frac{\Delta H^\ominus}{RT} - \frac{\Delta S^\ominus}{R} \quad (12)$$

where ΔH^\ominus is the standard enthalpy of reaction and ΔS^\ominus is the standard entropy of reaction.

From this analysis, the natural log of the equilibrium constant and the natural log of the hydrogen partial pressure are both linear functions of $1/T$. The enthalpy of formation is obtained from the slope of the curve, while the entropy of formation is obtained from its intercept.

For exothermic reactions, the enthalpy of formation is negative, which yields a negative slope for plots of $\ln(p_{H_2})$ vs. $1/T$ and a positive slope for plots of $\ln(K_{eq})$ vs. $1/T$.

4.2.1.4 Enthalpies and entropies of formation

The enthalpies and entropies of formation for yttrium dihydride are important to understand, as they provide information on how tightly bound the hydrogen is within the hydride moderator and, thus, helps describe the propensity of the moderator to release hydrogen at a given set of conditions.

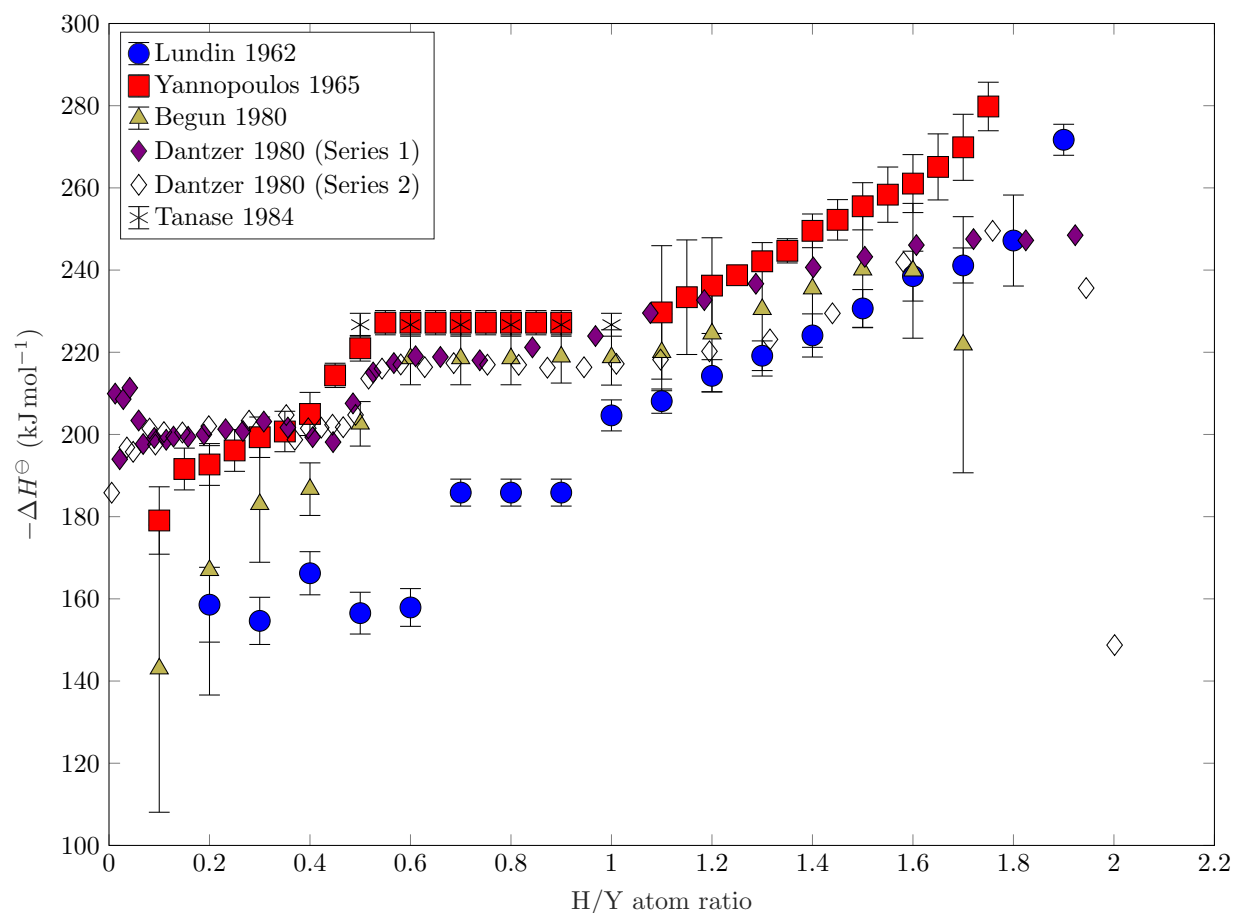


Figure 4.3: Partial molar enthalpy of formation for YH_x as a function of hydrogen content in H/Y units. Data taken from [12], [49], [54], [55], [64].

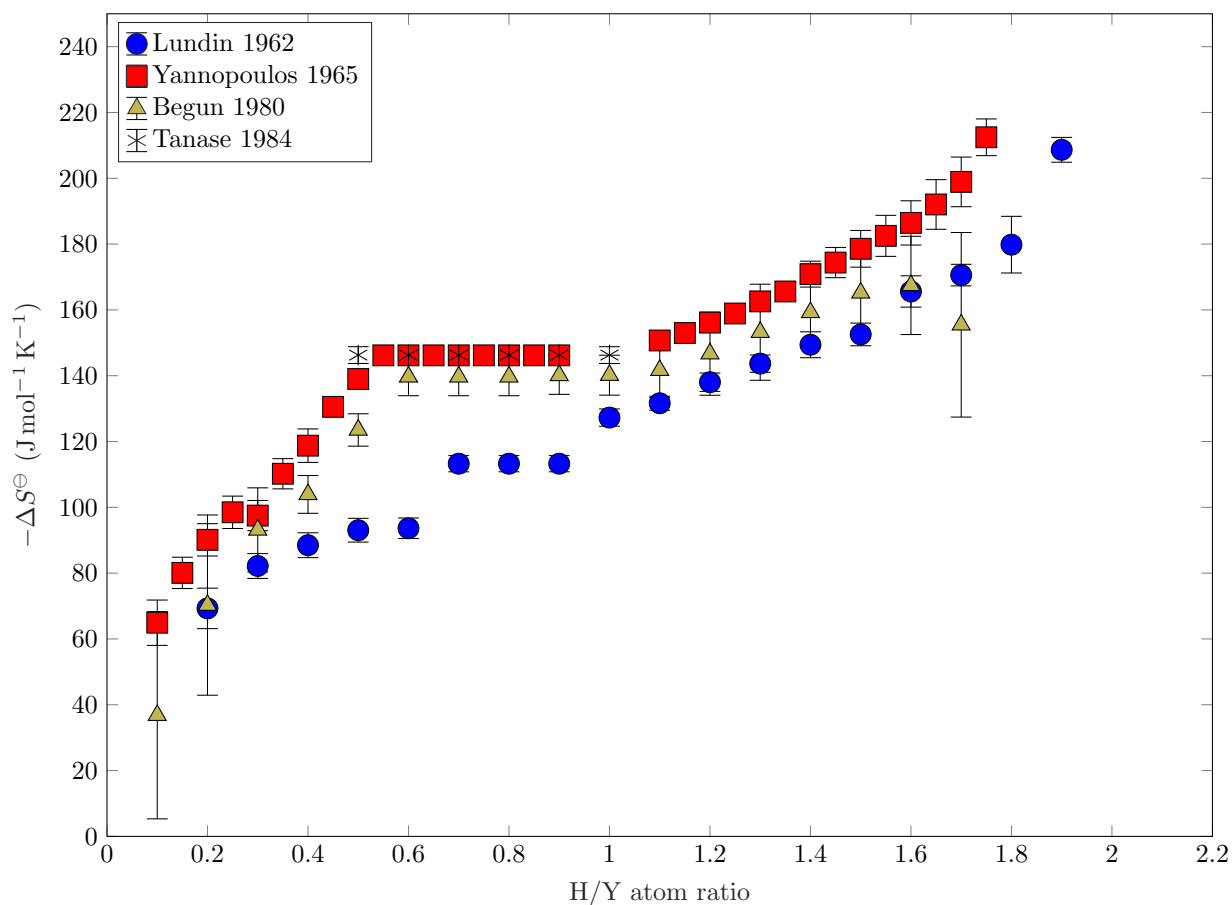


Figure 4.4: Partial molar non-configurational entropy of YH_x as a function of hydrogen content in H/Y units. Data taken from [12], [49], [54], [55].

The partial molar enthalpies of formation in the literature are plotted in Figure 4.3, while the partial molar entropies of formation are plotted in Figure 4.4. It should be noted that all results except those from Dantzer et al. were calculated using the above methods from the PCT curves, while Dantzer et al. measured the partial molar enthalpies of formation using calorimetric methods. Another available dataset by Fadeyev et al. was not included, due to a lack of access, as it is a foreign publication not available in English [66].

From Figure 4.3, it is observed that many of the data sets exhibit a significant degree of overlap. The only dataset that appears to disagree is the one by Lundin et al., for which the data appear to be consistently lower, as compared with the other literature. The temperature range examined in the study by Lundin et al. was higher than for all other studies, which could have impacted the analysis. In all studies, the enthalpies of formation are essentially constant in the two-phase equilibrium region of $\alpha\text{-Y} + \delta\text{-YH}_2$ for hydrogen contents between approximately 0.5 and 1.0 H/Y units. In the $\delta\text{-YH}_2$ single-phase region, the enthalpy of formation consistently increases with hydrogen content, though Dantzer et al. measured a significantly lower value at 2.0 H/Y units. However, in the $\alpha\text{-Y}$ single-phase region, the relationship between enthalpy of formation and hydrogen content is not as clear. The data by Dantzer et al. show a nearly constant enthalpy of formation in this region, while the values determined from other studies, calculated using the PCT curves, show increasing formation enthalpy with hydrogen content. It should be noted from Section 4.1.2.1 that there exists a significant degree of scatter in the PCT data for the $\alpha\text{-Y}$ single-phase region across all datasets.

As was the case for Figure 4.3, in Figure 4.4, it is observed that the datasets exhibit a significant degree of overlap, with only the data from Lundin et al. deviating from the others. Again, this is believed to be due

to the higher temperature at which hydrogen absorption experiments were performed. However, this is only the case for the α -Y single phase and α -Y + δ -YH₂ two-phase regions. In the δ -YH₂ single-phase regions, all datasets exhibit consistency. The non-configurational entropy of formation is observed to increase with hydrogen content up to approximately 0.5 H/Y units, which demarcates the α -Y and α -Y + δ -YH₂ regions. In the two-phase region, the entropy of formation is constant. Finally, in the δ -YH₂ single-phase region, the non-configurational entropy increases with hydrogen content.

4.2.2 Phase-formation kinetics

While hydriding kinetics have been measured for many materials such as titanium, zirconium, and intermetallic compounds for hydrogen storage applications, the formation kinetics of yttrium dihydride have not been explicitly measured [53], [67]. One study in the literature examined the emissivity of a sample of yttrium metal as a function of time when exposed to 0.49 MPa (4.8 atm) of hydrogen at 528 K (250 °C), though the result was unclear due to the formation of yttrium trihydride before complete reaction of yttrium metal to form the dihydride [68]. The application of methods described in literature may be used to characterize the formation of yttrium dihydride from yttrium metal at various temperatures and pressures.

In general, the pure metals that form hydrides are highly electropositive and readily form surface oxide layers. These surface oxides act as barriers for hydrogen diffusion from the gaseous phase to the metal. As hydrogen diffuses to the metal, it reacts with the metal to form the metal hydride, which is accompanied by the characteristic volume expansion associated with metal hydride formation as the hydride ingresses into the metal in a film-like manner [53].

At low temperatures, the solubility of hydrogen in the metal is low, which results in a large hydrogen concentration gradient across the surface [53]. Under these conditions, hydrogen accumulates below the surface oxide layer and, thus, the hydrides nucleate just below the surface oxide layer and grow into the metal. The solubility of hydrogen in the metal increases exponentially with temperature. Thus, at high temperatures, the hydrogen concentration gradient is shallower and, thus, allows for hydrogen to diffuse further within the metal due to a slower approach to supersaturation at the surface. As a result, other nucleation sites beside the oxide/metal interface become possible such as pathways for fast diffusion like grain boundaries. This results in bulk nucleation of hydrides.

The observed effect of temperature and diffusion pathways on hydriding kinetics makes it clear that surface condition, the presence of impurity phases, heat treatment, and physical form (powder vs. monolith) may affect hydriding kinetics significantly. Thus, future investigations into the hydriding kinetics of yttrium to form yttrium dihydride must account for these types of effects.

4.2.3 Heat capacity

Experimental measurements of the thermophysical properties of yttrium dihydride are limited. Heat capacity has been measured experimentally and new, DFT-based methods allow for simulation of this property. Flotow et al. measured the heat capacity at low temperatures (5 – 350 K) using an adiabatic calorimeter and cryostat [18]. Parker also reported heat capacity, as measured using an ice calorimeter, in the temperature range of 273 to 1081 K [16]. More recently, Ito et al.; Shivprasad et al.; and Trofimov et al. measured the heat capacity of yttrium dihydride using DSC [17], [19], [69]. Ito et al. and Trofimov et al. measured heat capacity for various stoichiometries of yttrium dihydride, while Shivprasad et al. measured the heat capacity of near-stoichiometric sintered samples and compared experimental results with those calculated using DFT. A summary of these measurements and calculations is shown in Figure 4.5. The heat capacities of end member stoichiometries from Trofimov et al. are plotted.

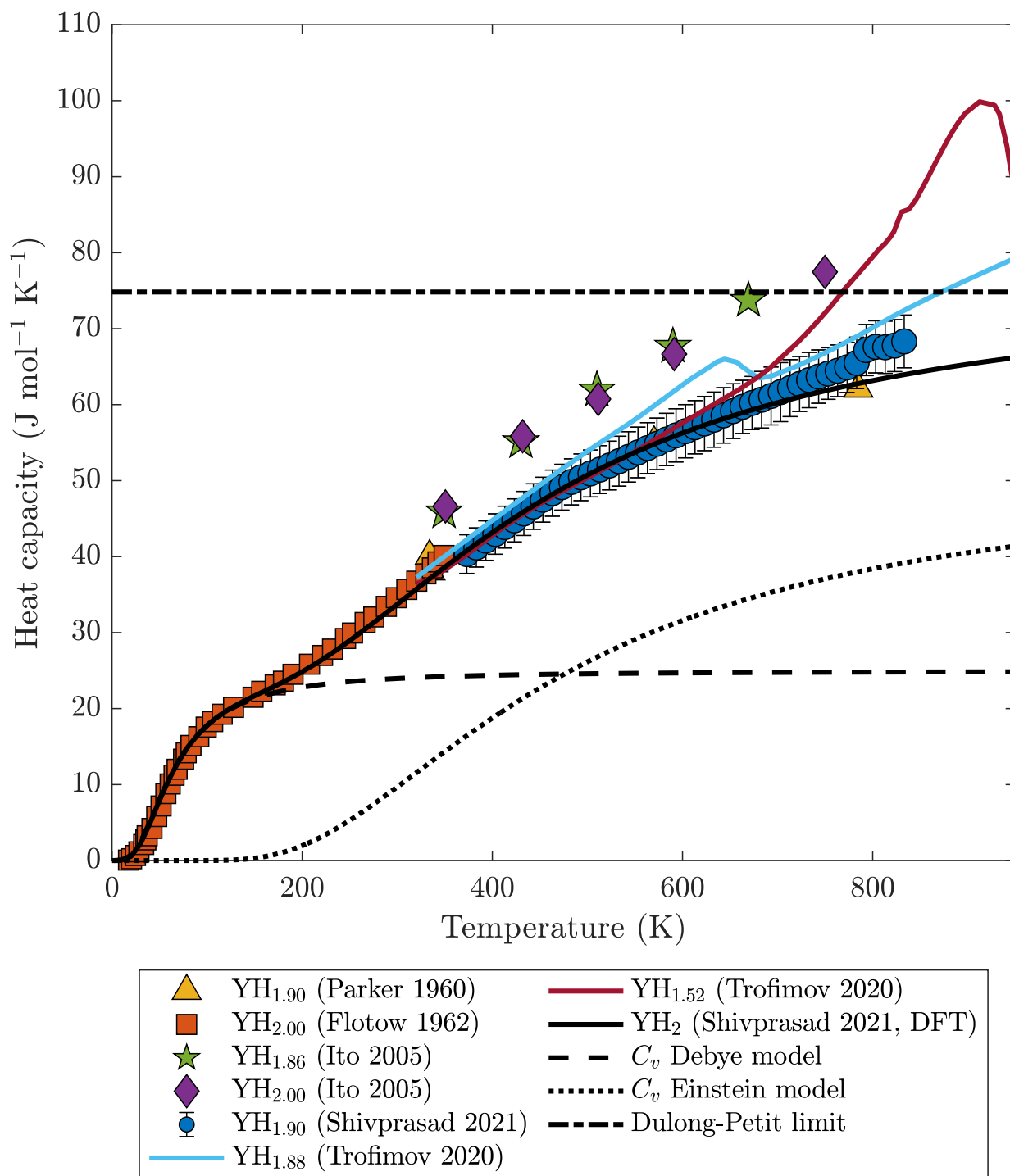


Figure 4.5: Molar heat capacity of yttrium dihydride as a function of temperature. Data from [16]–[19], [69]. The heat capacities of end member stoichiometries from Trofimov, et al. are plotted.

Figure 4.5 shows that there exists a significant degree of overlap between experimental data. Particularly, the data from Flotow et al., Parker, Shivprasad et al., and Trofimov et al. are consistent. The experimental data from Ito et al. show higher molar heat capacities, even exceeding the Dulong-Petit limit

for the heat capacity. It is believed that the data presented by Flotow et al., Parker, and Shivprasad et al. represent the more correct values.

It is also observed that the results of DFT modeling by Shivprasad et al. are also consistent with the experimental data up to a temperature of approximately 700 K (427 °C). Above this temperature, the modeling results and experimental data appear to diverge. This could be due to changes in hydrogen content experimentally from passing from the two-phase to the single-phase region or, as was hypothesized by Shivprasad et al., to the initial formation of Frenkel pair defects [17]. Trofimov et al. similarly hypothesized that the divergence, shown as a peak in the heat capacity data, was due to an order-disorder transition [69]. However, the concentration of gas impurities for each of the referenced studies is not known. As a result, more studies are required to understand the dependence of the molar heat capacity with temperature in this higher-temperature regime.

4.2.4 Thermal expansion and density

As with heat capacity measurements, thermal expansion data are similarly limited. Lundin et al. and Parker measured the thermal expansion coefficient of yttrium dihydride using dilatometric methods [15], [16]. More recently, Setoyama et al., Shivprasad et al., and Trofimov et al. measured thermal expansion using XRD, neutron diffraction, and dilatometry, respectively [14], [17], [69]. All thermal expansion coefficients show consistency in values and in functional behavior with temperature. The experiments by Lundin et al. and Shivprasad et al. measured data at high temperature, showing that at a temperature of 800 K, the dependence of the thermal expansion coefficient with temperature appeared to change. Shivprasad et al. hypothesized that this change may be due to the formation of Frenkel pair defects, as their formation is accompanied a volume expansion, which would be reflected in the thermal expansion.

Shivprasad et al. provided empirical fits for the two temperature regimes considered for thermal expansion as equations with the following form: $\alpha_p = A + BT + CT^2$. Hu et al. fit the TCR CTE data to a function with the following form: $\alpha = A + BT + CT^2 + DT^3$ [70]. Here, we have taken a similar approach as Hu et al: we have fit the CTE data across many studies using a third-order polynomial with temperature units centered and scaled to the span. These fits are shown for YH_{1.90} in Figure 4.6 and fitting parameters are summarized in Table 4.4. The CTE of YH_{1.61} from the TCR program is shown for comparison.

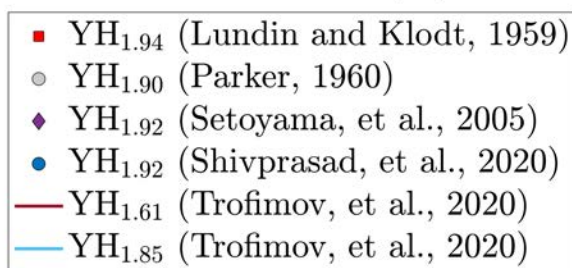
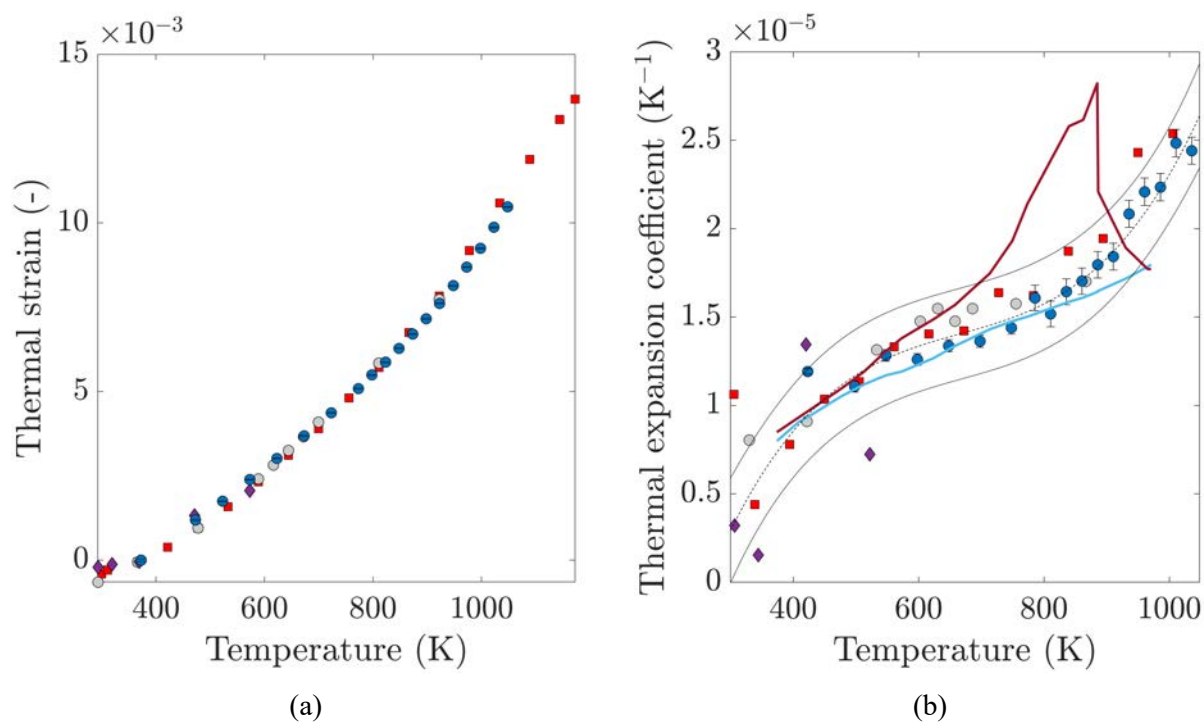


Figure 4.6: (a) Thermal strain and (b) thermal expansion coefficient for yttrium dihydride. Thermal expansion coefficient was fit to functions of temperature, the fitting parameters for which are summarized in Table 4.4. Figure adapted from [17].

Table 4.4: Empirical fitting parameters for thermal expansion coefficient as a function of temperature. Fitting parameters are given with 95% confidence intervals.

Temperature range (K)	Temperature unit (-)	A (10 ⁻⁵)	B (10 ⁻⁶)	C (10 ⁻⁷)	D (10 ⁻⁶)
298 – 1000	$\frac{T - 698.0}{203.6}$	1.44 ± 0.05	2.10 ± 0.76	6.13 ± 3.82	1.30 ± 0.38

From Figure 4.6(a), it is observed that the thermal strain is consistent across all the literature studies. Similarly, the thermal expansion coefficients are also consistent, with differences arising from the sensitivity of the relative techniques.

Shivprasad et al. correlated the shift in the thermal expansion coefficient to a peak in the heat capacity data, though the temperatures were slightly different. The authors provided possible explanations including crossing a phase boundary and the formation of Frenkel pair defects, both of which would change the molar volume and, thus, present as changes to the thermal strain and the coefficient of thermal expansion. Trofimov et al., also observed a similar correspondence between the hypothesized order-disorder transition across heat capacity and thermal expansion data.

4.2.5 Thermal conductivity

Three experimental measurements of the thermal conductivity of yttrium dihydride exist. Parker measured this property as a function of temperature for $\text{YH}_{1.77}$ using a heat flow method and apparatus [16]. Ito et al., Shivprasad et al., and Trofimov et al. calculated thermal conductivity from experimentally-measured thermal expansion coefficient, thermal diffusivity, and heat capacity [17], [19], [69]. Reported values from these studies are plotted in Figure 4.7, along with the calculated thermal resistivities. The results from Ito et al. and Shivprasad et al. appear to be consistent, while the values from Parker are much lower. Indeed, the values provided by Parker are closer to values expected for yttrium metal than yttrium dihydride. As a result, it is believed that the values by Ito et al. and Shivprasad et al. are the correct thermal conductivities.

Shivprasad et al. provided an empirical fit for the temperature range considered for thermal resistivity values; this is given in Equation (13). This fit also encompasses the data from Ito et al., due to the consistency of the experimental data between the two studies.

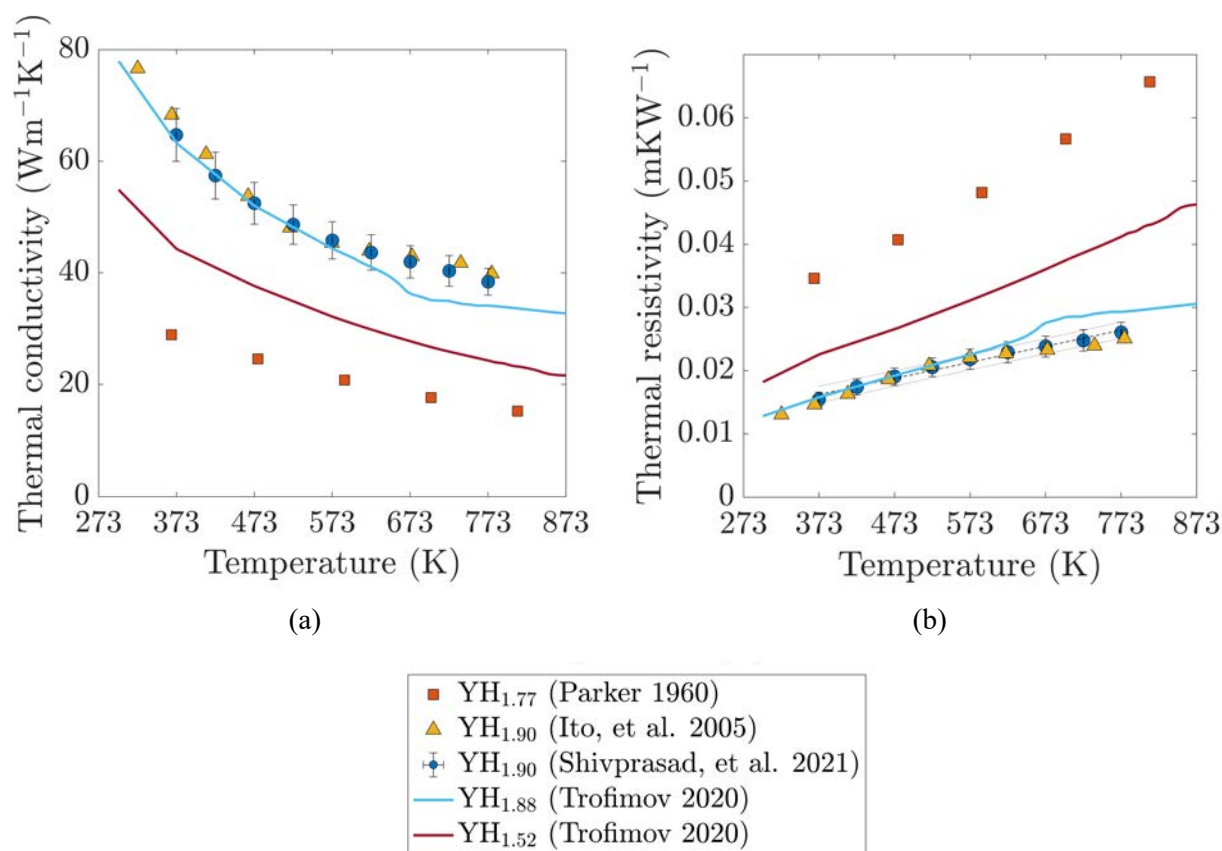


Figure 4.7: (a) Thermal conductivity and (b) thermal resistivity for yttrium dihydride as a function of temperature. Data from [16], [17], [19].

$$R_{\lambda} = (6.50 \times 10^{-3} \pm 1.70 \times 10^{-3}) + (2.61 \times 10^{-5} \pm 0.28 \times 10^{-5})T \quad (13)$$

4.2.6 Emissivity

The emissivities of yttrium hydrides have been measured qualitatively. The infrared emissivity of yttrium was measured as a function of exposure time when exposed to 0.49 MPa (4.8 atm) of pure hydrogen at 528 K (250 °C) [68]. Normalized emissivity values showed that emissivity increased with hydrogen content and exposure time. The highest emissivity was recorded for yttrium trihydride, with the emissivity of yttrium dihydride lower than for yttrium trihydride, and the emissivity of yttrium metal was lower still. It is difficult to extrapolate emissivities of yttrium dihydride from these measurements, as the phase contents did not appear to follow a strict $\alpha\text{-Y} \rightarrow \text{YH}_2 \rightarrow \text{YH}_3$ progression. Rather, by 10 minutes, the phase mixture consisted of $\alpha\text{-Y}$ and yttrium dihydride, after which point yttrium trihydride began to form, indicating that the system consisted of two to three phases at any point when yttrium dihydride was present.

4.2.7 Electrical properties

Electrical property measurements have been performed for yttrium hydrides because of their use in switchable mirror applications. Because electrical properties do not significantly impact nuclear reactor performance, a cursory summary is presented here.

Weaver et al. measured the dielectric constant of yttrium dihydride as approximately 4.8 [71]. This value increases for higher concentrations, with the dielectric coefficient for yttrium trihydride of 10 [72]. This is consistent with the understanding that the dihydride is well known as having metallic character, while the trihydride is a semiconductor [73]. The charge carrier concentration of yttrium dihydride was measured by Sakai et al. as between 1.1 and $1.4 \times 10^{27} \text{ m}^{-3}$ [33]. Sakai et al. also measured the electrical resistivity of yttrium dihydride as approximately $8.4 \times 10^{-12} \text{ } \Omega\text{m}$. With higher stoichiometries in the two-phase region (YH_2+YH_3), Vajda et al. measured a significant increase in electrical resistivity. For $\text{H/Y}=2.065$, the electrical resistivity was measured as $8.4 \times 10^{-7} \text{ } \Omega\text{m}$ at 283 K and for $\text{H/Y}=2.07$, the electrical resistivity was measured as approximately $1.85 \times 10^{-6} \text{ } \Omega\text{m}$ at 256 K [74]. For stoichiometries close to (but above) $\text{H/Y} = 2.00$, the hydride was found to be a metal at low temperatures, but still transitioned to a semiconductor at temperatures below room temperature [75], [76]. This transition temperature was found to decrease as a function of increased hydrogen content. These data have been evaluated for validity by other sources [77]–[80].

4.2.8 Magnetic properties

Magnetic property measurements of yttrium dihydride have been summarized more thoroughly in other reviews [81]. Since they do not significantly impact nuclear reactor operation, a qualitative summary is presented here. Arons summarized that, in general, the rare earth dihydrides exhibit an antiferritic structure, but that this structure changes ordering with hydrogen stoichiometry [81]. Additionally, because measured hydrogen contents are relatively inaccurate, true dependence of the magnetic properties with hydrogen content are difficult. Magnetic properties, including proton spin relaxation times were found to depend on the impurity content with even 100 ppm of rare earth additions significantly impacting these values [21]. It was also observed that different additions affected the relaxation times to differing degrees, with the purest yttrium having the longest relaxation time and relaxation time decreased with additions in this order: cerium, dysprosium, erbium, neodymium, gadolinium. Because the proton spin relaxation time measurement is used to determine hydrogen self-diffusion coefficients, it is believed that diffusion measurements of hydrogen in yttrium dihydride may have been significantly impacted by impurities. A more detailed description of the NMR technique and the different relaxation times is given in Section 4.3, as it relates to the determination of hydrogen self-diffusion parameters through yttrium dihydride.

4.2.9 Microstructure

The microstructure of as-hydrided yttrium has not been standardized for nuclear applications. Nevertheless, the yttrium hydride microstructure from the ANP program is presented in the literature. These data show the effect of increasing hydrogen content on the phase fraction and hydride grain morphology [43]. At low hydrogen concentrations, ($\text{H/Y} < 0.3$, approximately), all hydrogen is in solid solution in the yttrium metal; at a slightly higher concentration, above the solid solubility limit, the hydride phase grains precipitate inside and around the H-saturated yttrium grains. With increasing hydrogen concentration, the phase fraction of the hydride phase increases. At a hydrogen concentration of approximately 1.15 H/Y units, the system is a 50:50 phase mixture of H-saturated yttrium metal and yttrium dihydride. At a hydrogen concentration at, or near, stoichiometry ($\text{H/Y} \sim 2$), the phase fraction of yttrium metal is extremely low, and the metal has nearly completely converted to the dihydride. More recent examinations through the TCR program note the effect of hydrogen absorption rate on microstructure wherein it was noted that high H_2 flowrates lead to large concentration gradients and therefore high internal stress and subsequent twinning. When the H_2 flow rate was low, however, the concentration gradients were small such that a randomly oriented grain structure formed [70]. We note that the yttrium metal used in the TCR program contained a fluorine impurity and precipitated as fluoride inclusions in the hydride bulk. The effects of these inclusions on physical properties remains an open question.

4.2.10 Mechanical properties

Until recently, mechanical testing data of yttrium dihydride have been relatively limited, primarily due to the difficulty in fabricating bulk, crack-free YH_{2-x} specimens. Nevertheless, the limited studies on elastic properties and hardness are presented below.

Short-term tensile testing of yttrium dihydride was performed as part of the ANP program, though the program noted the difficulty in performing these tests, due to the brittle nature of the material and high temperature testing was performed in hydrogen-containing atmospheres to maintain stoichiometry [16]. Briefly summarized here, the results of short-term tensile testing showed nearly zero elongation at temperatures up to approximately 1000 °C but were able to measure up to 5.5% elongation at nearly 1200 °C and an 18% reduction in area at the same temperature. The author noted that, for near-stoichiometric yttrium dihydride, the yield strength could not be measured and the variation in the ultimate tensile strength was nearly 100% of the measurement. The final, published result is given in Reference [16].

Equibiaxial flexural testing of yttrium dihydride was performed under the TCR program for geometries representative of the moderator for the TCR design. No significant trend was observed as a function of hydrogen content in the range of 1.00 to 1.87 H/Y units, though the mean failure strength was calculated to range between 50 and 95 MPa for 5.8 mm diameter samples with a nominal thickness between 0.5 and 0.625 mm [70].

The elastic moduli of yttrium dihydride are commonly determined through sound speed or resonant ultrasound spectroscopy (RUS) measurements. Beattie et al. measured the elastic moduli of $\text{YH}_{1.93}$ using sound speed measurements in the temperature range 80-300 K and observed an inflection around 230 K possibly due to proton ordering on tetrahedral sites [82]. Setoyama et al. measured the moduli of various yttrium dihydride specimens as a function of hydrogen content ($\text{H/Y} \sim 1.6$ to 2.3), finding that the moduli increase linearly with H/Y [14], [83]. Similarly, Hu et al. observed a linear dependence of the elastic moduli with H/Y [84], [85]. Shivprasad et al. demonstrated that the moduli of sintered-yttrium dihydride monoliths decrease linearly as a function of monolith porosity (90-100%) [2]. We also note there exists a significant body of work done to calculate the elastic moduli using DFT with modeled results in significant agreement with experimental values for fully-dense material [2], [34]–[36]. The results of various elastic modulus measurements are plotted in Figure 4.8 as a function of stoichiometry as well as in Figure 4.9 as a function of porosity.

Results for elastic moduli as a function of hydrogen content were fit to linear functions of hydrogen-to-yttrium atom ratio and are summarized as follows with 95% confidence intervals for the fitting parameters:

$$G(\text{GPa}) = (20.0 \pm 8.9) + (18.1 \pm 4.8)c_H \quad (14)$$

$$K(\text{GPa}) = (21.0 \pm 27.4) + (34.1 \pm 14.8)c_H \quad (15)$$

$$E(\text{GPa}) = (16.1 \pm 33.6) + (61.7 \pm 18.2)c_H \quad (16)$$

It should be noted that the confidence intervals for the bulk and Young's moduli are significantly larger than for the shear modulus; Shivprasad et al. showed, many of the resonances required to measure the elastic moduli have predominantly shear characteristics [2]. That is, material resonances from which the elastic moduli are calculated predominantly depend on shear modes. As a result, further characterization of the elastic moduli to stoichiometries lower than $\text{H/Y} = 1.90$ are required to better characterize the dependence of these moduli on hydrogen content.

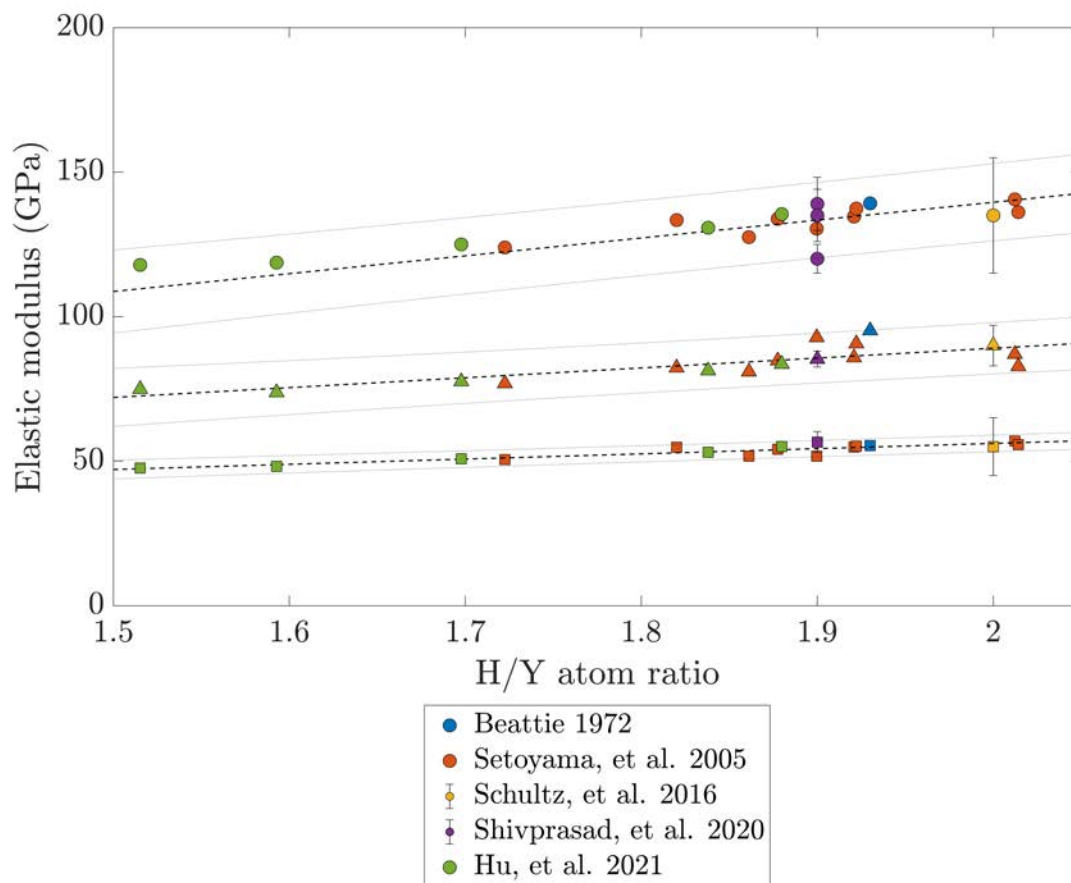


Figure 4.8: Shear, bulk, and Young's moduli as a function of hydrogen content for yttrium dihydride. Data from [2], [14], [36], [83], [85]. Young's moduli are plotted in circles, bulk moduli are plotted in triangles, and shear moduli are plotted in squares.

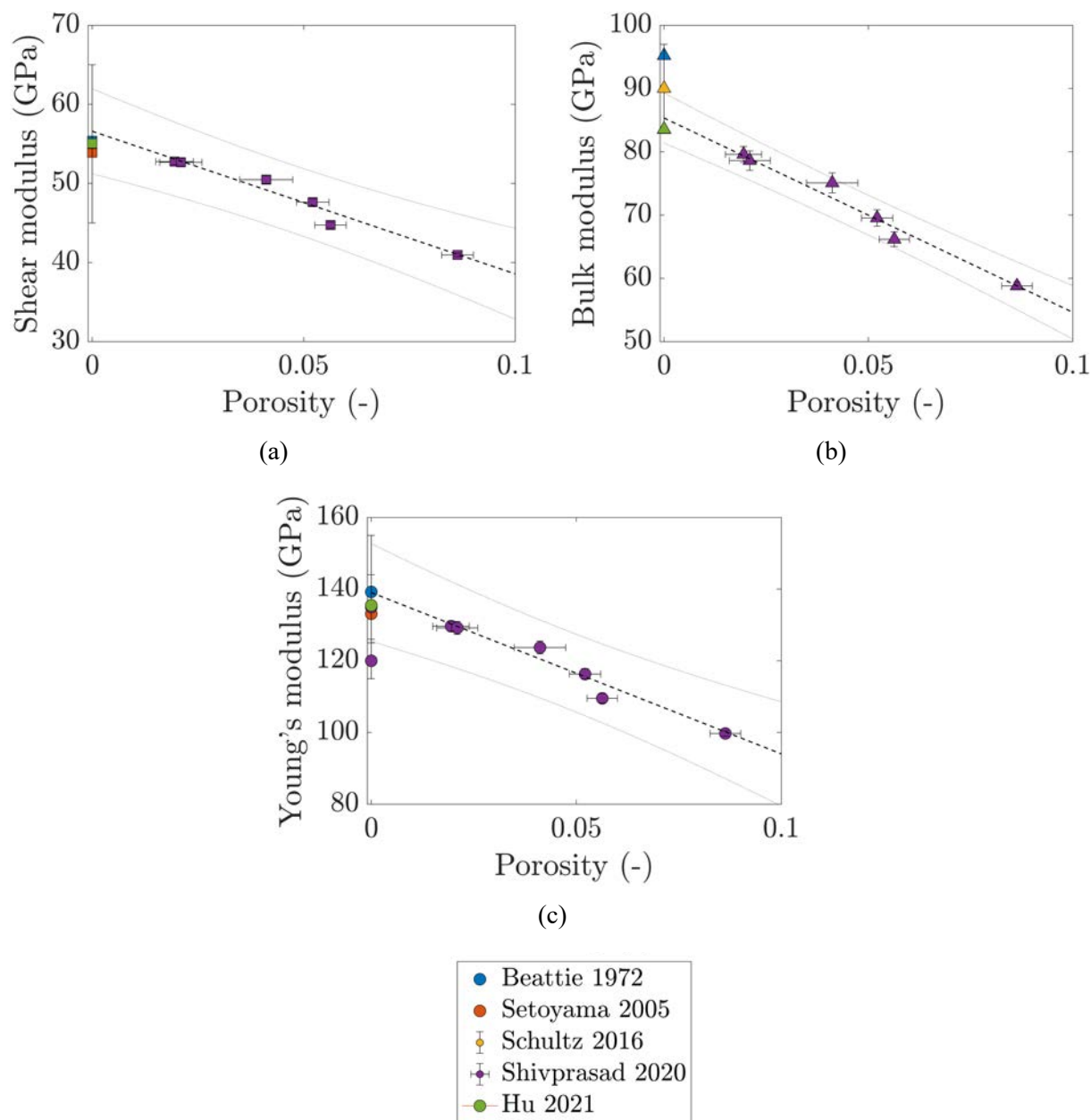


Figure 4.9: (a) Shear, (b) bulk, and (c) Young's moduli of yttrium dihydride as a function of porosity. Data from [2], [14], [36], [83], [85].

From Figure 4.9, it is apparent that the elastic moduli of yttrium dihydride decrease with increasing porosity. It is also observed that the dependence on porosity extrapolates to elastic moduli values very close to values obtained from fully dense yttrium dihydride. As a result, it is believed that these values and fits are valid. These results have been fit as functions of porosity and are given as follows with 95% confidence intervals for the fitting parameters:

$$G(\text{GPa}) = (56.60 \pm 3.64) - (180.34 \pm 70.81)p \quad (17)$$

$$K(\text{GPa}) = (85.34 \pm 2.68) - (307.12 \pm 52.03)p \quad (18)$$

$$E(\text{GPa}) = (139.02 \pm 9.23) - (449.88 \pm 179.44)p \quad (19)$$

Detailed studies on hardness of yttrium dihydride are limited. Funston, Parker, Setoyama et al., and Hu et al. measured hardness as a function of hydrogen content, while Shivprasad et al. measured hardness on yttrium dihydride fabricated by direct-hydriding and powder metallurgical methods [2], [14], [16], [85], [86]. Vickers hardness results are summarized in Table 4.5.

Table 4.5: Summary of hardness values for yttrium dihydride. Data from [2], [14], [16], [85], [86].

Source (-)	Vickers hardness (GPa)
YH _{1.4} [86]	2.0
YH _{2.12} [16]	2.9
YH _x (1.7 ≤ H/Y ≤ 2.0) [14]	3.48 ± 0.86
YH _{1.90} (sintered) [2]	2.87 ± 0.69
YH _{1.90} (direct hydride) [2]	3.17 ± 0.07
YH _{1.88} [85]	2.39 ± 0.08

Funston, Parker, and Shivprasad et al. examined the hardness of only one stoichiometry of yttrium dihydride, while Setoyama, et al. and Hu, et al. looked at the dependence of the Vickers hardness on hydrogen content. While Setoyama et al. noted a dependence on hydrogen content, the values for H/Y between 1.7 and 1.9 exhibited no significant variation. Conversely, Hu et al. observed a linear trend for H/Y values between 1.5 and 1.9, but their reported values were lower than in other studies. As a result, we believe that further study of hardness as a function of hydrogen content is warranted.

4.3 Hydrogen self-diffusion in yttrium hydrides

Although the data are sparse, hydrogen diffusion measurements in yttrium hydrides have been performed using nuclear magnetic resonance (NMR) and quasielastic neutron scattering (QENS). A brief overview of both techniques is given below to familiarize the reader, which is followed by a summary of relevant diffusion parameters of YH_{2-x}.

NMR. ¹H-NMR spectroscopy is a non-destructive, bulk measurement technique that measures the precession of hydrogen nuclei and provides insight into chemical composition, spin-spin coupling between adjacent atoms, molecular dynamics, and molecular structure of hydrogen-bearing media. The NMR signal is sensitive to the electronic structure since the number and proximity of nearby electrons affect the local magnetic field at the nucleus. Each nucleus precesses at a unique frequency ω , which is defined as

$$\omega = \gamma B_0 \quad (20)$$

where B_0 is the applied magnetic field and γ is the gyromagnetic ratio, which is characteristic of each nucleus and is representative of the sensitivity of a nucleus in an NMR experiment [87]. During a single measurement, only one nuclide can be observed, as the spectrometer can only detect signal at one frequency.

When a given sample is placed in a strong magnetic field, the nuclear spin aligns with the magnetic field along the z-axis, as is shown in Figure 4.10(a). A radiofrequency (RF) pulse is then applied, which perturbs the spin, aligning it either perpendicular to (x-y plane) or anti-parallel (-z-axis) to the magnetic field, as shown in Figure 4.10(b). Relaxation causes the nuclear spin to be aligned again with the magnetic

field along the z-axis (Figure 4.10(c)–(e)) and causes a free induction decay, which is measured and Fourier transformed to give a spectrum. There are two mechanisms of relaxation: spin-lattice relaxation, which is also referred to as longitudinal or T_1 relaxation, and spin-spin relaxation, referred to as transverse or T_2 relaxation [88].

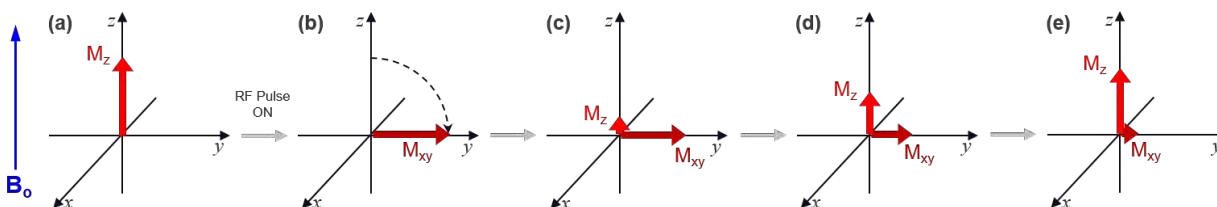


Figure 4.10: Magnetization evolution before and after a pulse of RF radiation is applied. The applied magnetic field is in the direction of the z-axis.

The T_1 relaxation is the recovery of the magnetization along the z-axis and is illustrated in Figure 4.10. This relaxation is typically caused by coupling to the electronic spin of free electrons within the material (i.e., dangling bonds, paramagnetic impurities, dopants, impurities). In pure, crystalline materials, the T_1 relaxation is very long and can take minutes or even hours for full recovery of the magnetization to the z-axis [88]. The T_2 relaxation, on the other hand, is more complex and may be affected by many factors. One such cause will be explained simply here for the sake of brevity. The first is related to the precession of the nuclear spin and is depicted in Figure 4.11: After the RF pulse is turned off, the magnetization is aligned along the +y-axis, and the nuclear spins begin to precess about the z-axis. If we consider here ^1H nuclei in the tetrahedral site in yttrium dihydride, all ^1H nuclei should be precessing at the same frequency. However, some of the hydrogen atoms may be in close proximity to a vacancy or an impurity atom, which disturbs the local magnetic field, B_{loc} [88]. The frequency then becomes:

$$\omega = \gamma(B_0 + B_{loc}) \quad (21)$$

There is then a loss in phase coherence between the unaffected spins and the affected spins. After some time, the ^1H nuclei are resonating at their own frequency, which is due to the differences in the local magnetic field [88]. An analogy for this phenomenon is that of runners in a marathon. Before the race, all runners are lined up, and initially after the start, all competitors are running at the same speed. After time t , some of the runners will be going at a faster pace. Towards the end of the race, the runners are randomly dispersed throughout the racetrack due to their various speeds. Another factor that leads to loss of phase coherence is mobility of the nuclei since hopping from one site to another will drastically change the local magnetic field, and this will lead to a change in the T_2 relative to the case of non-mobile species.

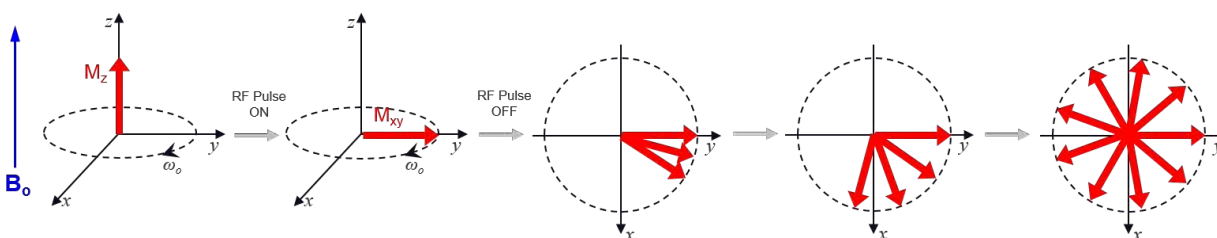


Figure 4.11: Loss of coherence of magnetization in the x-y plane after an RF pulse is applied.

The T_2 relaxation time can be approximated as the inverse of the width of the spectral peak. Therefore, if the nuclei in yttrium dihydride become mobile, the T_2 will be affected, thus changing the peak width of the observed signal. This allows the study of material dynamics since the mobility of a species can be related to the T_2 . The dynamics can be studied by solid-state, variable temperature (VT) NMR and has been applied broadly to determine the rotational dynamics in ionic liquids and glasses [89], [90], the glass transition temperature of glasses and hopping transport in ionic conductors and cathode materials [91], [92].

QENS. Quasielastic neutron scattering is an inelastic neutron scattering technique that measures the very small energy exchanges between incident neutrons and the sample's nuclei, which in turn leads to broadening of the spectrum's elastic line. The probability of a scattering event is gauged by the scattering cross sections of the sample's nuclei. In this context, the scattering cross section of hydrogen (~82 barns) is much greater than that of zirconium (~6 barns), and so the scattered neutron signal is dominated by the dynamics of hydrogen. The measured energy exchange occurs in the sub-meV range, which corresponds to dynamic processes such as hydrogen self-diffusion in the pico-to-microsecond range. The reader is referred to the book by Marc Bée for a detailed description of the QENS technique [93].

Results. Hydrogen diffusion in the hydrogen sublattice of the δ -phase hydride is believed to proceed via three mechanisms: (1) tetrahedral-tetrahedral (T-T) jumps, tetrahedral-octahedral (T-O) jumps, and octahedral-octahedral (O-O) jumps. As mentioned above, the δ -phase hydride forms when the tetrahedral sites of the crystal fill with hydrogen, while the ε -phase hydride forms upon accommodation of further hydrogen into the octahedral sites. However, as noted above, octahedral site occupation has been observed even at room temperature [20], [26].

The mechanism for hydrogen diffusion in yttrium hydrides have been observed to depend on hydrogen content and temperature [20]. At low hydrogen contents, the hydrogen diffusion mechanism is hypothesized to be predominantly due to T-T jumps. As the hydrogen content increases, the tetrahedral sites become increasingly occupied, which allows contributions from T-O and O-O jumps to the diffusion mechanism. It is hypothesized that the T-O and O-O jumps have lower activation energies, as the overall diffusion activation energy has been observed to decrease with increasing hydrogen content.

Table 4.6 summarizes key hydrogen diffusion data, including activation energy (E_A) and pre-exponential factor (D_0), for yttrium hydrides as a function of hydrogen content. For stoichiometries where pre-exponential factors were not available in the literature, they were approximated using the Einstein equation for diffusion:

$$D_0 = \frac{1}{6} z f v \lambda^2 \quad (22)$$

where z is the number of nearest-neighbor diffusion sites (6 for T-T jumps), λ is the distance to the nearest-neighbor site (half the lattice parameter for T-T jumps), and f is a correction factor of approximately 0.74 [94].

Table 4.6: Available diffusion parameters for the Y-H system. D_0 values were calculated as described in the text when not provided in the reference.

H/Y	E_a (eV)	D_0 (cm^2/s)
1.63 [20]	1.1	5.3×10^{-1}
1.80 [26]	0.3	n/a
1.87 [27]	1.73	45.7
1.91 [95]	0.53	4.8×10^{-4}

1.92 [20]	0.438	7.9×10^{-5}
1.95 [95]	0.4	9.0×10^{-5}
1.98 [20]	0.417	7.9×10^{-5}
2.03 [95]	0.38	1.0×10^{-4}

4.4 Degradation of un-irradiated yttrium dihydride

This section represents a significant gap in our current knowledge regarding yttrium dihydride performance as a moderator. It is currently hypothesized that the material will degrade under the thermal cycling that will occur during startup/shutdown cycles for a nuclear reactor. Phase stability will be affected by significant temperature changes under hydrogen partial pressures such that yttrium trihydride may be stable at low temperature, which could result in the degradation of thermophysical and mechanical properties, as well as pulverization of the moderator. Fast temperature changes could result in thermal shock of the moderator. Additionally, operation of the moderator at high temperatures (e.g., above 500 °C) could allow for hydrogen loss through a moderator cladding, reactor monolith, or pressure vessel. However, the relevant testing parameters have not been critically evaluated. As such, thermo-mechanical testing for this type of behavior has yet to occur.

In general, it was established early on that metal hydride moderators would undergo hydrogen losses due to temperature and hydrogen redistribution under temperature gradients, as observed in the ANP and SNAP programs [96]–[98]. To mitigate hydrogen losses, a permeation-resistant cladding is necessary. Historic concepts focused on FeCrAl alloys with native aluminum oxide layers and ceramic coatings to the cladding inner diameter of Hastelloy-N in the ANP and SNAP programs, respectively.

In the event of a reactor breach, the ingress of air will enable degradation of the moderator via reactions with nitrogen and oxygen. The ability of yttrium compounds to readily react with oxygen is well-documented in the literature, as yttrium is found at the bottom of the Ellingham diagrams for oxide formation [99]. However, yttrium dihydride has also been observed to react at room temperature with nitrogen to form surface layers of yttrium nitride. Nitrogen rot of yttrium dihydride has also been observed in the past, though no details of this phenomenon have been provided in the literature other than the fact that it may be impeded by grinding off the surface oxide/nitride and re-oxidizing the surface [100].

A recent examination of yttrium dihydride oxidation in synthetic air by Kane et al. showed that yttrium dihydride oxidized more slowly than yttrium metal [101]. The authors attributed the difference to proton migration from the hydride into the oxide scale, thus inhibiting oxygen diffusion from the oxide/air interface to the oxide/hydride interface. It was noted that the presence of yttrium fluoride (YF₃) and yttrium oxy-fluoride (YOF) impurities in the base metal carried over to the hydride phase and increased the oxidation rate, but this effect was reduced in the hydride as compared to yttrium metal.

Hu et al. performed thermal desorption experiments of yttrium dihydride and found that the hydrogen desorption process occurred in four distinct regimes [85]. In the first regime (I), hydrogen losses were minimal, while in regime II, hydrogen desorption flux was approximately linear with temperature. In regime III, hydrogen flux was constant with temperature, and in regime IV, the hydrogen desorption flux exhibited a temperature dependence.

5. NEUTRONIC CONSIDERATIONS OF YTTRIUM DIHYDRIDE

The figures of merit for the neutronic performance of moderators are the moderating power and the moderating ratio [102]. The moderating power is defined as the logarithmic energy decrement per collision (also called lethargy), ξ , multiplied by the macroscopic neutron scattering cross-section, Σ_s . The moderating ratio is defined as the ratio of the moderating power to the macroscopic neutron absorption cross-section,

Σ_a . Both parameters are necessary for moderator qualification, as the former describes the efficiency with which neutrons are slowed down, while the latter describes the ability to moderate with minimal neutron absorption.

It should be noted that materials with very high moderating power may have low moderating ratios due to high absorption cross-sections. For example, gadolinium dihydride, GdH_2 , has a very high hydrogen density at room temperature (7.43×10^{22} atoms/cm³ [47]), but a very low moderating ratio because it also has a very high neutron absorption cross-section [45]. Conversely, some materials may have a very low moderating power but a very high moderating ratio. For example, beryllium metal, Be, has a very low moderating power because it is relatively massive for a moderator material, but has a very high moderating ratio because it has a very low neutron absorption cross-section.

5.1 Lethargy and moderating power

The lethargy is defined as the average effectiveness of a substance in slowing down neutrons for all collisions based on an assumption of equal probability for all collision angles between 0° and 180° [102]. For a neutron colliding with a nucleus of mass, A , then the equation for the lethargy is:

$$\xi = 1 + \frac{(A-1)^2}{2A} \ln \frac{A-1}{A+1} \quad (23)$$

For example, the lethargy for a beryllium-moderated system would be:

$$\xi_{Be} = 1 + \frac{(9-1)^2}{2 \times 9} \ln \frac{9-1}{9+1} \approx 0.2066$$

Equation (23) may also be approximated as

$$\xi = \frac{2}{A + 2/3} \quad (24)$$

The approximation in Equation (24) is good for values of A greater than 10 but is still quite accurate for low values of A . For $A = 1$ (H-1 nucleus), $\xi = 1$.

Lethargy may also be used to calculate the average number of collisions required to slow down a neutron from one energy to another, \bar{n} , which is defined as:

$$\bar{n} = \frac{\ln(E_0/E)}{\xi} \quad (25)$$

For example, the average number of collisions required to thermalize 1 MeV neutrons using beryllium metal would be

$$\bar{n}_{Be} = \frac{\ln(E_0/E_{th})}{\xi_{Be}} = \frac{\ln \left[\frac{1 \times 10^6}{0.025} \right]}{0.2066} = 84.66 \approx 85$$

Thus, beryllium metal requires, on average, 85 collisions to thermalize a neutron.

Values of ξ and \bar{n} for various nuclei relevant to moderator materials are summarized in Table 5.1.

Table 5.1: Lethargy and average number of collisions required to thermalize 1 MeV neutrons for various nuclei of interest for nuclear reactor moderator applications.

Nucleus	Mass number	ξ	\bar{n} for 1MeV to 0.0253 eV
H-1	1	1.0000	17
H-2	2	0.7253	24
He-4	4	0.4253	41
Be-9	9	0.2066	85
C-12	12	0.1578	111
O-16	16	0.1199	146

Table 5.1 shows that H-1 nuclei are the most efficient at thermalizing neutrons, with approximately 17 collisions required to thermalize 1 MeV neutrons, on average. As the mass number of a nucleus increases, the number of collisions drastically increases.

The moderating power, P , of a material is defined as:

$$P = \sum_i \xi_i \Sigma_{s,i} = \sum_i \xi_i N_i \sigma_{s,i} \quad (26)$$

where N_i and $\sigma_{s,i}$ are the atomic density and scattering cross-section of nucleus, i , respectively. Because of the dependence of Equation (26) on lethargy and scattering terms, only, moderating power may be thought of as a moderating efficiency. That is, neutrons will slow down faster (with fewer collisions) using moderators with higher P -values.

The purpose of slowing neutrons is to obtain thermal neutrons from fast neutrons. Thus, the appropriate scattering cross-sections to use are in the epithermal regime [102]. In general, elastic scattering is considered as the primary mechanism for energy loss via collisions except for nuclei with very high atomic numbers.

The atomic density, N_i , may be calculated readily from the mass density and the stoichiometry. For all materials, the mass density changes as a function of temperature. Given a reference density, ρ_i^0 , at a reference temperature, T_0 , and a known coefficient of thermal expansion, α_i , the mass density as a function of temperature, $\rho_i(T)$, is calculated as [103]:

$$\rho_i(T) = \frac{\rho_i^0}{[1 + \alpha_i(T - T_0)]^3} \quad (27)$$

For metal hydrides, the equilibrium hydrogen composition also changes as a function of temperature [2]. Metal hydrides also have lattice parameter changes with stoichiometry [14]. However, the lattice parameter changes with H/M atom ratio contribute much less to overall atom density than does the stoichiometry change with temperature.

Moderating power as a function of temperature in 1 atm of pure hydrogen for some materials considered for nuclear reactor moderator applications is presented in Figure 5.1.

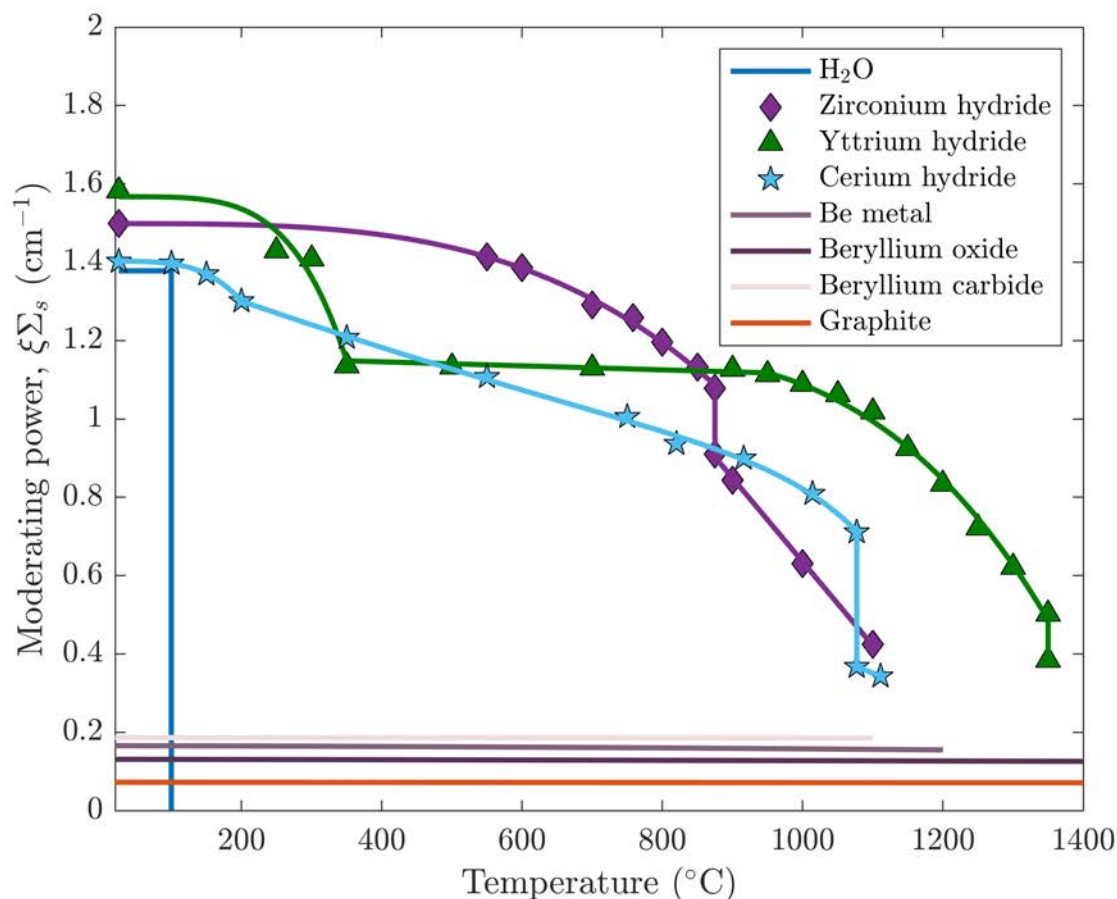


Figure 5.1: Moderating power as a function of temperature for various candidate moderator materials. All data points are taken from literature for materials at 1 atm pressure [12]–[14], [47], [49], [104]–[116]. Data for metal hydrides were determined for those materials under 1 atm of pure hydrogen.

Figure 5.1 plots the moderating power as a function of temperature for various candidate moderator materials including water, metal hydrides, beryllium compounds, and graphite. It is observed that the moderating power of water and the metal hydrides is significantly higher than for the beryllium compounds and graphite. This makes sense, as the primary moderating nucleus in water and metal hydrides is hydrogen, which is much more efficient at slowing down neutrons than any other nucleus. As a result, metal hydrides and water are more efficient at thermalizing neutrons than are the beryllium-based compounds and graphite. However, one advantage of beryllium-containing compounds and graphite is that their moderating power is nearly constant over temperature with only small changes due to thermal expansion. Conversely, metal hydrides may have large changes in moderating power with temperature due to changes in equilibrium hydrogen content as a function of temperature. It is for this reason that yttrium dihydride is attractive as a metal hydride moderator, as its hydrogen content is relatively constant from approximately 350 °C to 900 °C in 1 atm of pure hydrogen, where zirconium hydride and cerium hydride lose hydrogen with temperature in this regime. However, maintaining a 1 atm hydrogen pressure is optimistic. In most realistic scenarios, hydrogen dissociation will occur.

5.2 Moderating ratio

From Equation (26), it is observed that the moderating power does not account for the loss of neutrons due to neutron absorption. A simple parameter to account for neutron absorption is the macroscopic neutron absorption cross-section, Σ_a . Combining the moderating power with the macroscopic absorption cross-section results in the moderating ratio, R , as follows:

$$R = \frac{P}{\Sigma_a} = \frac{\sum_i \xi_i N_i \sigma_{s,i}}{\sum_i N_i \sigma_{a,i}} \quad (28)$$

where $\sigma_{a,i}$ is the microscopic absorption cross-section of nucleus, i .

For a moderator with the chemical formula: $A_a B_b C_c \dots Z_z$, the density of each atomic specie, K , is given as follows:

$$N_K = \eta_K \frac{\rho_t N_A}{M_t}$$

where η_K is the number of atoms of specie K in the chemical compound for the moderator material, ρ_t is the total mass density of the compound, M_t is the total molar mass of the compound, and N_A is Avogadro's number.

Thus:

$$\begin{aligned} P &= \frac{\rho_t N_A}{M_t} \sum_i \xi_i \eta_i \sigma_{s,i} \\ \Sigma_a &= \frac{\rho_t N_A}{M_t} \sum_i \eta_i \sigma_{a,i} \\ R &= \frac{\sum_i \xi_i \eta_i \sigma_{s,i}}{\sum_i \eta_i \sigma_{a,i}} \end{aligned} \quad (29)$$

As seen in Equation (29), because the macroscopic cross-sections are present in the numerator and denominator of the moderating ratio, the dependence of the moderating ratio on the atomic density disappears. This means that the moderating ratio provides information on the moderating ability of a chemical compound and ignores density. For example, the moderating ratio of liquid water and steam are both 62.11, although liquid water is a much better moderator, due to its higher density of hydrogen atoms (over three orders of magnitude higher than that of steam). As another example, an ideal gas of hydrogen has a very low atomic density. As a result, its moderating power is very low ($5 \times 10^{-4} \text{ cm}^{-1}$) but its moderating ratio is very high (61.45) because it is 100% hydrogen. Some moderating ratios for some candidate moderator materials at room temperature are summarized in Table 5.2.

Table 5.2: Moderating lethargy, powers, and ratios for various candidate moderator materials at room temperature and ambient pressure.

Material	ξ	$P = \xi \Sigma_s$ (cm ⁻¹)	$R = P/\Sigma_a$
H ₂ (gas)	1.0000	5×10^{-4}	61.45
H ₂ O (liq)	0.7066	1.38	62.11
ZrH _{2.0} (solid)	0.6739	1.66	37.45
YH _{2.0} (solid)	0.6741	1.22	17.32
CeH _{2.0} (solid)	0.6714	0.94	22.94
Be (solid)	0.2066	0.17	133.99
BeO (solid)	0.1633	0.13	173.99
Be ₂ C (solid)	0.1903	0.19	149.15
Graphite (solid)	0.1578	0.07	202.30

From Table 5.2, it is seen that the materials with high hydrogen density tend to have high moderating power, but low moderating ratio due to neutron absorption by other atoms in the compounds. Conversely, the beryllium-containing and carbon-containing moderator compounds have very low moderating powers, but very high moderating ratios, due to the small neutron absorption cross-sections of beryllium-9, oxygen-16, and carbon-12.

The compounds shown here are the primary candidate moderator materials of interest. Beryllium and its compounds are well-characterized as reflector materials. In this report, they will be discussed in terms of technical challenges associated with application as a moderator. Graphite is similarly well-characterized as a moderator and a discussion of its moderator performance will be briefly discussed. However, metal hydride moderators are the main focus of this report, specifically the exploration of alternative alloy hydrides that may be able to overcome the technical challenges associated with pure metal hydrides.

5.3 Effect of hydrogen diffusion on neutronic properties

As mentioned in Section 2.1.1., the issue with hydrided zirconium as a moderator is that above 500°C it begins to lose substantial amounts of hydrogen [104]; these loss rates are further increased with temperature [117], [118]. The latter could be detrimental for reliable reactor operation, and thus the use of hydrided zirconium is limited to lower-temperature applications. Yttrium dihydride had been tested as a higher temperature moderator during the 1960s as part of the aircraft propulsion project [5], [12], [15], [43]. Yttrium dihydride was studied, but never fully developed for use in a working reactor system. Yet it was recognized even then that yttrium dihydride could be used at temperatures up to 800°C with no significant change in the hydrogen density.

One of the challenges of the metal hydrides, is that under a temperature gradient, hydrogen redistributes within its adjacent metallic bond and migrates under the influence of a temperature gradient (see Figure 5.2). This weakness has been a target of critical controversies as to whether one should utilize metal hydrides in reactors due to the inherent non-linearity in reactor dynamics resulting from hydrogen migration. Simply put, when hydrogen migrates inside a reactor core, it leads to fluctuations in the behavior and performance of the system. The hydrogen relocates within the yttrium hydride moderator leading to local changes in material properties. As such, the neutronic, mechanical/structural, and heat transfer physics are directly affected. In terms of neutronic effects, local fast and thermal cross sections are interrupted due

to changes in the availability of hydrogen. The shift in hydrogen would lead to changes in thermomechanical properties, which, in turn, would lead to deviations in the temperature and material stress profile. Asymmetric geometrical changes in materials and evolving temperature profiles change the microscopic and macroscopic neutron cross sections, which develop into further neutronic complications. Note that during this convoluted physical process, overall actinide and non-actinide inventory of the fuel still needs to be accounted for proper neutronics modeling. All the details described above result in a shift of the local and global neutron energy spectra, and thus, effective neutron multiplication of the system, which affects criticality. One can easily imagine how non-linear and complicated it can become to design a hydride moderated reactor without accounting for the hydrogen migration complications and the fluctuations within the reactor system from the beginning until the end of reactor life.

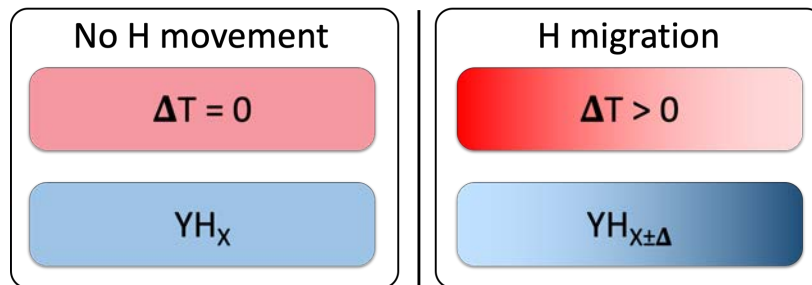


Figure 5.2: Hydrogen migration under the influence of temperature gradient.

To tackle this highly “turbulent” problem, one must take meticulous, well directed steps to make hydride moderated systems a reality for today’s applications. Results from neutron diffraction experiments carried out at Los Alamos Neutron Science Center combined with high fidelity atomic scale modeling and coupled Multiphysics codes allowed the creation of new thermal scattering cross sections for a range of yttrium hydride stoichiometries [17]. The overall fundamental behavior of yttrium dihydride properties is described below.

Lower hydrogen concentration tends to decrease the heat capacity of yttrium dihydride due to lower availability of hydrogen. Lower stoichiometry (H/Y ratio) introduces phonon splitting in the yttrium dihydride crystal, which in turn, ends up increasing the thermal scattering properties, $S(\alpha,\beta)$, per atom for certain energies. Higher hydrogen concentration tends to make the yttrium dihydride material less ductile, thus causing it to be more likely to crack, break, or pulverize. On one hand, more hydrogen content in a reactor is beneficial from neutron stopping power perspective. On the other hand, more hydrogen could compromise the structural integrity of moderator material in the reactor. As such, hydrogen hotspots in the moderator material can lead to severe material deformation and/or failure. Higher temperatures in hydride favor higher hydrogen diffusion rates, thus driving the reactor to be more dynamic, more chaotic, and therefore less controllable. A summary of the effects of hydrogen content on various thermodynamic, physical, and neutronic properties is given in Table 5.3.

Table 5.3: Summary of the effect of hydrogen content on material and neutronic properties.

Less hydrogen	Property	More hydrogen
Higher	Mass density	Lower
Lower	Hydrogen density	Higher
Higher	Resistance to dehydriding	Lower
Lower	Heat capacity	Higher
Lower	Thermal conductivity	Higher
Lower	Elastic moduli	Higher
Higher	Ductility	Lower
Lower	S(α,β)	Higher
Lower	Moderating power	Higher
Lower	Moderating Ratio	Higher

6. HISTORICAL IRRADIATION OF YTTRIUM DIHYDRIDE IN FFTF

Due to its properties, yttrium dihydride was selected as the high temperature moderator for the Fast Flux Test Facility (FFTF) for various anticipated missions related to the production of beneficial isotopes. Yttrium dihydride moderator was irradiated in the FFTF for ~138 EFPD during Cycle 9A (September 1986) in the Cobalt Test [119], for ~10 EFPD during Cycle 11A (May 1989) for the Multiple Isotope Irradiation (MIP) Test [120], for ~185 EFPD in the Materials Open Test Assembly (MOTA)-1D (August 1985) [121] and for ~203 EFPD in the fusion MOTA-2B (May 1991) [122]. In addition, yttrium dihydride was selected for the high temperature moderator for the Space Isotope Production (SIP) core in FFTF for producing ^{238}Pu with low ^{236}Pu for NASA from the irradiation of ^{237}Np . In preparing for these various missions, in-reactor and out-of-reactor tests were completed that assessed the properties of yttrium dihydride, such as the hydrogen to yttrium ratio, the hydrogen diffusion and permeation rates, the effect of microstructure on hydriding, the effect of impurities on performance, and the effects of the hydriding procedure on the material properties. These were integral performance tests of specific configurations rather than basic materials properties tests.

The QA of the FFTF tests was equivalent to NQA-1. For FFTF irradiations and laboratory tests, the hydrogen content in yttrium dihydride pellets was measured by a scale that determined the weight change from the hydriding process. The requirements were that the weight had to change by the amount expected from the hydriding process, and the yttrium dihydride had to pass through a go-no go diameter gage. Bowing of the ~5 cm long pellets was checked by the gage, and some machining of the pellets was allowed so that the pellet could fit through the gage. The cladding was exposed to the sodium and was relied on for structural integrity. The temperature of the yttrium dihydride was calculated for the proposed configuration and materials, and as long as the calculated yttrium dihydride temperature was below the value that would result in high hydrogen partial pressure, then the test could proceed, as the hydrogen pressure would not be exceeded. Coatings to reduce hydrogen permeation from free hydrogen were used on the inside and outside of the steel cladding tubes for the Cobalt Test and some MOTA tests, but later tests (such as the MIP test) did not have these coatings, and no coatings were proposed for the SIP project.

No post irradiation information for yttrium dihydride from these FFTF tests (such as phase diagrams) was located.

7. POST-IRRADIATION EXAMINATION OF YTTRIUM DIHYDRIDE IRRADIATED IN THE ATR

7.1 Introduction to the irradiation experiment

Identifying a suitable moderator material is one of the primary challenges in the current small, high-temperature microreactor designs. Metal hydrides have large equilibrium hydrogen concentrations which make them particularly suitable for reactor moderator applications. However, the hydrogen retention at elevated temperatures is a challenge for the microreactor designs. Yttrium dihydride is particularly attractive for high temperature reactors because it offers a hydrogen containment capability exceeding that of other metal hydrides. Initial research into the possibility of using metal hydrides as reflector and moderator materials can be traced back to the 1950s era nuclear aircraft programs. From the well-known phase behavior, yttrium dihydride has considerably better hydrogen retention than the zirconium hydrides at elevated temperatures suitable for microreactor applications. The hydrogen retention under irradiation conditions, however, is not well studied. Additionally, the physical integrity of the yttrium dihydride under irradiation has not been adequately investigated. Further, optimization of the yttrium dihydride fabrication method (whether direct hydriding or powder metallurgy) requires a relative comparison of the resultant thermophysical properties between the two methods. To investigate these areas, an ATR irradiation experiment (LANL-MOD-1) has been designed and fabricated to provide data on thermal and mechanical material properties, material stability and physical degradation, and hydrogen retention and diffusion as it relates to moderator worth in microreactor designs. LANL-MOD-1 is a non-fueled drop-in experiment in support of investigating yttrium hydride as a high temperature moderator material for microreactors.

LANL-MOD-1 will be irradiated in early 2021 in the ATR Northeast lobe, small B2 position, its relative position in ATR shown in Figure 7.1. The test train is vertically stacked with six titanium-zirconium-molybdenum (TZM) inner capsules containing the yttrium dihydride specimens. Three separate ATR fixture baskets are used with different gas mixtures to obtain three target material temperatures of 600, 700, and 800 °C, as shown in Table 7.1.

Table 7.1. Specimen test matrix showing the three target temperatures, fabrication methods, and characterization techniques for PIE analysis.

Sample Type	Nominal Temp	RUS	LFA	DSC	TEM	GDOES
(-)	(°C)	(-)	(-)	(-)	(-)	(-)
Direct	600	3	2	6	1	4
PM	600	3	2	6	1	4
Direct	700	3	2	6	1	4
PM	700	3	2	6	1	4
Direct	≤800	3	2	6	1	4
PM	≤800	3	2	6	1	4
Total	-	18	12	36	6	24

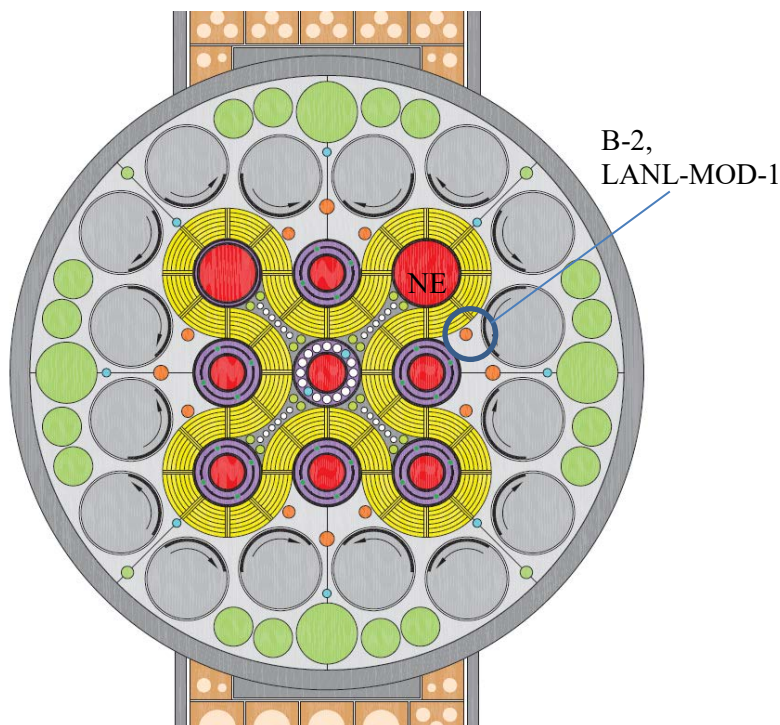


Figure 7.1: ATR core showing LANL-MOD-1 in the B2 position.

The cylindrical yttrium dihydride samples will be housed in a TZM inner capsule cladding and placed in ATR fixtures that also contain melt and fluence wires. These ATR fixtures will be contained within a sealed stainless-steel basket containing a helium/argon mixed atmosphere which will be inserted into the B2 position. Each ATR fixture will contain two of the inner TZM capsules, one containing yttrium dihydride produced by direct hydriding, and one containing yttrium dihydride produced by a powder metallurgy method. Figure 7.2 shows an elevated stack up of the six inner capsules in the B2 ATR position. The total fluence for a nominal 60-day ATR cycle is calculated to be 5×10^{20} n/cm²s, with small variations depending on specific specimen location within the vertical stack. This fluence level was chosen to allow material performance determinations at near expected lifetime fluence in microreactor operation. The TZM inner capsules are over-pressured with 10 torr (1333 Pa) of hydrogen, necessary to minimize hydrogen diffusion at elevated temperatures.

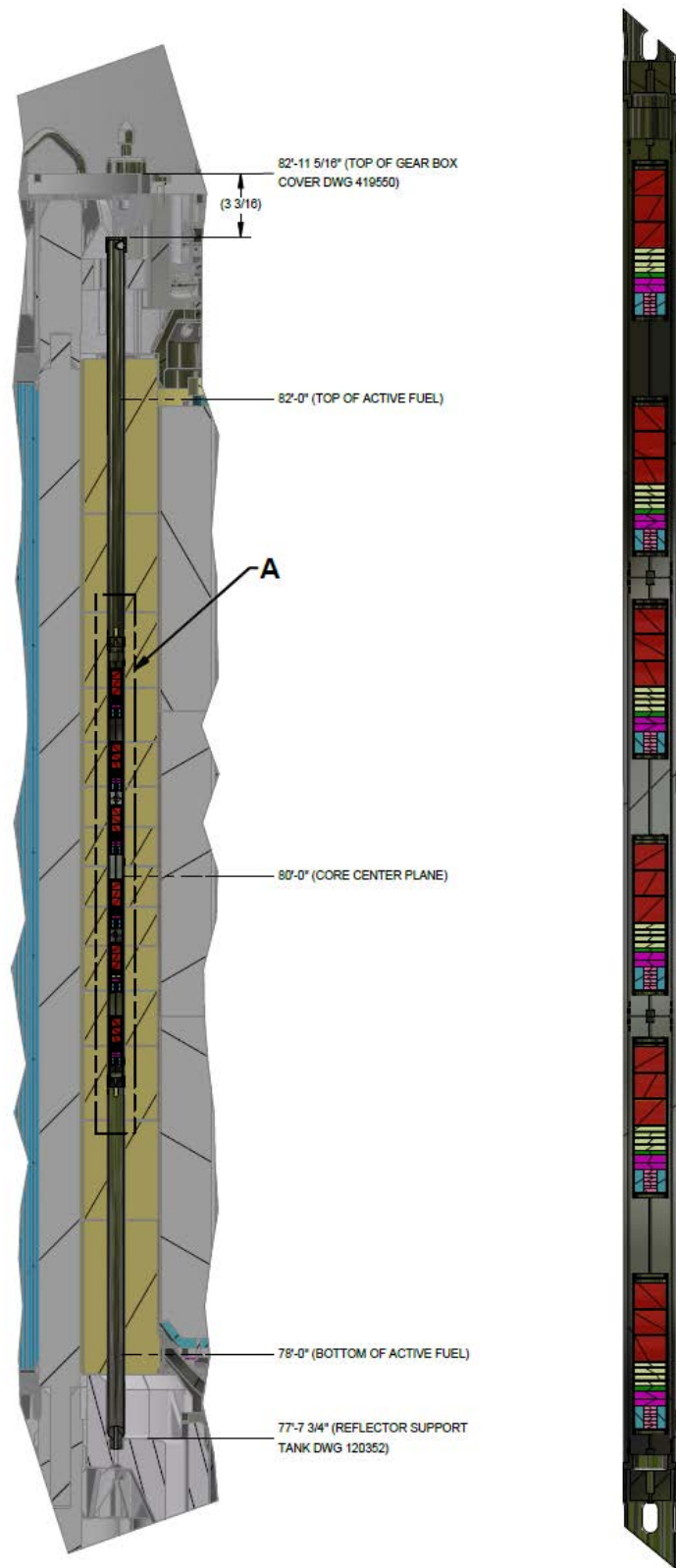


Figure 7.2. Elevation stack-up of the TZM capsules in the ATR B2 position.

7.2 Materials and Methods

As stated in the previous section, two types of yttrium dihydride are being evaluated in this irradiation experiment, a material produced by direct hydriding of bulk yttrium, and material produced by a powder metallurgy technique. The direct hydride method uses a standard Sievert's apparatus to slowly add hydrogen at temperature as described in Section 3.2. The hydride produced by powder metallurgy started from raw yttrium dihydride material that was ball-milled into fine powder before being compressed and sintered into green pellets as described previously. For both types of fabrication, the following cylindrical specimens were produced and identified by their intended PIE use:

RUS – Resonant ultrasound spectroscopy (12.5mm diameter x 10mm thick)

LFA – Laser flash analysis (12.5mm diameter x 2mm thick)

DSC – Differential scanning calorimetry (5mm diameter x 1.5mm thick)

TEM – Transmission electron microscopy (12.5mm diameter x 2mm thick)

GDOES – Glow discharge optical emission spectroscopy (12.5mm diameter x 2mm thick)

Additionally, thin TZM sheets were placed between the yttrium dihydride specimens to investigate clad-material interactions and hydrogen diffusion into the TZM. Again, the complete specimen test matrix is given in Table 7.1.

All the yttrium hydride material was produced with a $\text{YH}_{1.8 \pm 0.1}$ stoichiometry with the following physical and material properties:

Table 7.2. Yttrium hydride physical properties.

Property	Unit	Value
CTE	$\mu\text{m}/\text{m}\cdot\text{K}$	10.0
Thermal Conductivity	$\text{W}/\text{m}\cdot\text{K}$	15.0
Hardness	GPa	3.0
Density	gm/cm^3	4.26

Table 7.3. Total quantities of yttrium hydride for ATR PIE.

Sample ID (-)	Density (g/cm ³)	Volume (cm ³)	Mass per sample (g)	# of Samples (-)	Volume (cm ³)	Mass (g)
RUS	4.26	1.2272	5.2279	18	22.089	94.10
LFA	4.26	0.2454	1.0454	12	2.9448	12.545
DSC	4.26	0.0295	0.12567	36	1.062	4.524
GDOES	4.26	0.2454	1.0454	24	5.8896	25.10
TEM	4.26	0.2454	1.0454	6	1.4724	6.272
Total	-	-	-	96	33.5078	142.54 ^(*)

A single TZM inner capsule contains a complete set of experiment specimens. The TZM inner capsule is over pressured with 10 torr of hydrogen to minimize diffusion at higher temperatures. Two TZM inner capsules fit into a single ATR fixture basket, which also contains the passive fluence and temperature monitors, shown in Figure 7.3.

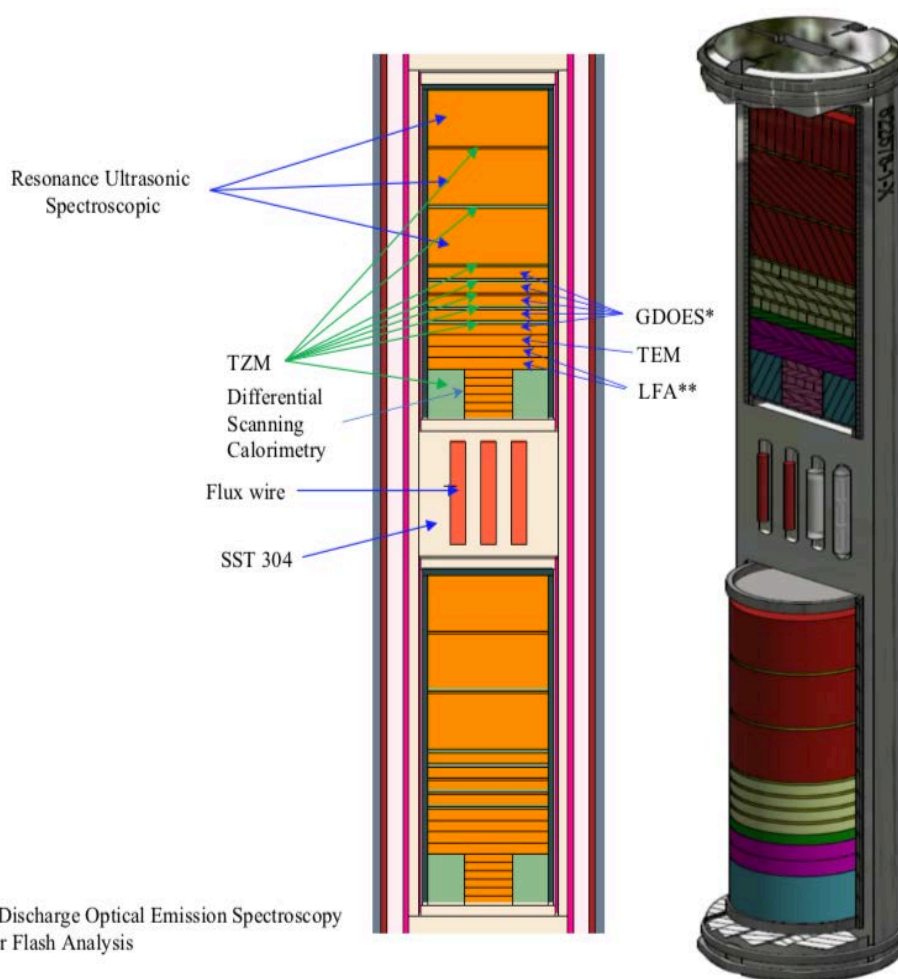


Figure 7.3. Hydride specimens located in a single ATR fixture basket. Two TZM inner capsules containing the hydride fit within a single ATR fixture, along with passive temperature and fluence monitors.

7.3 PIE results

Results of the post irradiation measurements will not be available until the irradiated capsules are removed and disassembled in the HFEF hot cell. Following irradiation, the capsules will be punctured to capture and analyze the gas spectrum to determine the extent of hydrogen loss from the yttrium dihydride. Any hydrogen loss from the yttrium dihydride would have a negative impact on the moderating worth of the material for use in microreactor application. Since hydrogen diffusion is so important to the material use, a number of confirmatory and different measurements are planned to accurately assess and understand the hydrogen retention.

Detailed mass balance and dimensional inspection will be performed on the yttrium dihydride specimens. Although a very simplistic technique, mass balance has been used with high reliability and accuracy in past hydride moderator studies. Dimensional analysis will indicate the overall physical integrity of the specimens and indicate any irradiation-induced swelling or cracking.

Specific specimen sizes and geometry were selected with the end state PIE in mind. Upon disassembly and obtaining the gas analysis, mass balance, and physical dilatometry for the specimens, they will be removed from the hot cell and distributed to the applicable measurement laboratories. RUS specimens will be used to measure the bulk elastic tensor for the material. This allows a direct comparison with the RUS performed on the unirradiated material. The LFA and DSC specimens will be used to measure the thermal diffusivity and thermal conductivity in a laser flash analysis. DSC samples will be used for differential scanning calorimetry to detect any potential phase change behavior in the specimens. The glow discharge optical emission spectroscopy will be used to look at hydrogen permeation and diffusion through the sample bulk. Additionally, microstructure of the specimens will be characterized to understand that diffusion in more detail.

8. INTEGRAL CRITICAL EXPERIMENT AT NCERC

8.1 Introduction to the critical experiment

As discussed above, a significant effort is being put forward to determine the material properties and differential cross sections for yttrium dihydride. This work also includes the determination of $S(\alpha,\beta)$ cross sections for yttrium dihydride in ENDF/B-VIII.0. The nuclear data pipeline does not end with the evaluation of the differential measurements – integral experiments are needed to ensure the nuclear data evaluations based on physics and differential measurements agree with reality; that is, they predict an accurate result of real systems. Thus, an integral experiment was designed to validate the cross-section behavior in a critical system. This includes not only absolute k_{eff} values, but also the reactivity feedback associated with non-nuclear heating. The designed experiment meets these purposes.

The Hypatia measurement campaign with yttrium dihydride moderators and highly enriched uranium (HEU) was completed in January 2021 at the U.S. Department of Energy's National Criticality Experiments Research Center at the Nevada National Security Site [123]. This measurement campaign provided unique integral measurements based on two experimental configurations and investigated the temperature effects of yttrium dihydride ($H/Y = 1.8$ and 1.9) in a critical reactor system. The Hypatia experiment consisted of a fuel column composed of HEU, 93 wt.% U-235 discs, yttrium hydride canned in titanium-zirconium-molybdenum (TZM), aluminum oxide heater plates, and other moderator and reflector materials (beryllium, depleted uranium, and graphite) inserted into a thick beryllium reflector. Thermal and neutronic calculations predicted that yttrium dihydride is a unique material that can exhibit a positive temperature coefficient of reactivity (i.e., reactivity can increase as the temperature in the hydride increases). Reactors using yttrium dihydride should account for this unique feature during design, and the results of the Hypatia experiment significantly aid that process.

8.2 Materials and methods

Most materials used in this experiment come from existing inventory at NCERC, highlighting its importance as a proof-of-concept and feasibility study. The existing materials include the radial reflector, the SNM (HEU), the axial reflectors, and most structural materials including the critical assembly machine, Planet. All aspects were held to standard QA procedures and level of rigor associated with other integral experiments at NCERC. A benchmark evaluation was not planned for these measurements, so a graded approach was taken for uncertainty minimization with a goal of balancing time and cost without compromising the most important aspects of the experiment. The design met all required criteria for any experiment performed at NCERC, including procurement of components which could affect the *Planet* shutdown mechanisms.

The overall layout is shown in Figure 8.1 below.

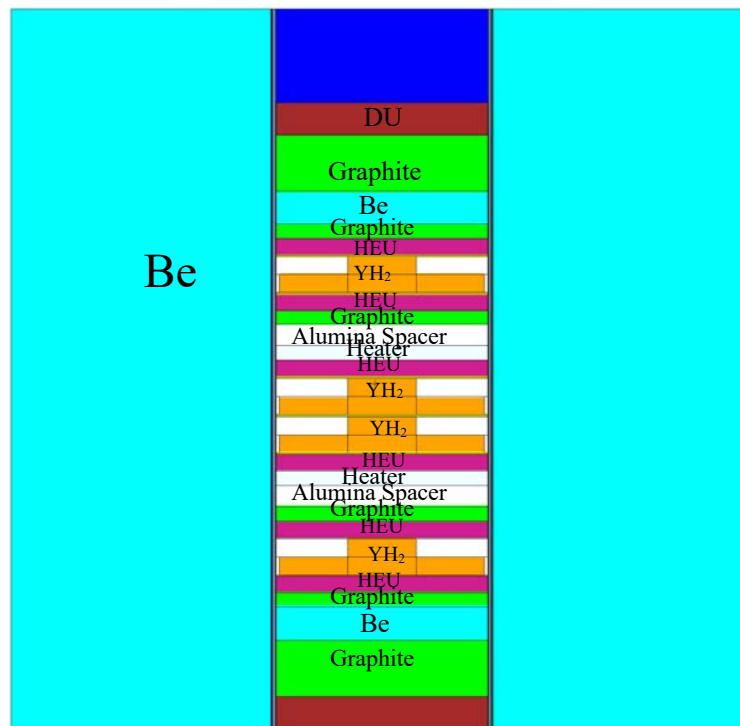


Figure 8.1. Critical experiment design overview.

Planet is a general-purpose critical assembly machine, which dates to the 1950s. It has undergone several changes since that time, including moving from LANL to Nevada in the 2000s. Most recently, *Planet* had upgrades to level all portions for perfect alignment.

The SNM is HEU metal discs, known as the C-discs, which are unclad. There are six in total with an average mass of 3920 g and average enrichment of 93.1 wt.% ²³⁵U. These discs are 5.94 inches in diameter and 0.465 inches tall. A picture of one is shown in Figure 8.2.



Figure 8.2. C-Disc (HEU Unalloyed Metal).

The axial reflectors included Be and DU discs, with diameters and height of 5.93 and 0.92 inches each, respectively.

Electric heaters were procured from NASA. These electric heaters allowed for only a central region to be heated (which included HEU and yttrium dihydride), thus minimizing competing effects. These heaters were manufactured to be slim and using materials without the potential to complicate the results. The heaters were a graphite coil with alumina shells on top and bottom. A diagram is shown in Figure 8.3.

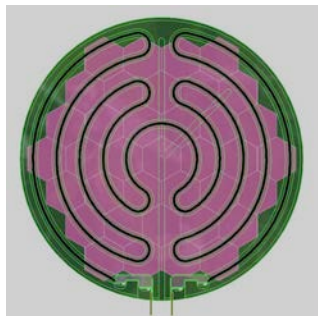


Figure 8.3. Alumina and graphite heater to be used in integral critical experiments at NCERC.

The actual experiments will take place as two-part series. The first will setup exactly as defined above, and the second will be similar but without the yttrium dihydride in the central heated region. Both series will be performed in the same manner. The experimental results will be defined in terms of reactor kinetics parameters. For example, the reactor period, τ , will be measured. This parameter defines the time it takes for the neutron population to increase by a factor e . At NCERC, this is measured by He-3 pulse counting tubes at low power, and by compensated-ion chambers at medium power. The reactor period is converted into reactivity in terms of how close the system is to super-prompt critical. The resulting excess reactivity value lies between 0 and 100 cents. At NCERC, we aim for 60 cents or less of excess reactivity and are regulated to less than 80 cents of excess reactivity.

1. A critical configuration, with 40-60 cents excess reactivity, will be obtained at room temperature by varying alumina spacer thickness and radial Be reflector height.
2. The core will be removed from the reflector and the temperature of the central heated region increased by 20-50 °C.
3. The core will be reinserted into the reflector and reactivity measured.
4. The process of Steps 2 and 3 will be repeated until either the maximum temperature (330 °C) is reached, or the system becomes subcritical.

8.3 Integral critical experiment results

The experiments will give the reactivity changes that occur in two similar systems with increasing temperature. The resulting changes will be compared with the predicted values from simulations with MCNP. The temperature is increased via electrical heaters to remove competing effects from nuclear heating. The systems, as discussed in Section 8.2 above, differ in that one does not have yttrium dihydride discs in the central heated region. A critical configuration, with approximately 40-60 cents excess reactivity, will be found for each configuration by varying radial beryllium height and alumina spacer height. The temperature will be increased in 20-50 °C increments from room temperature to 330 °C. At each point, the core will be inserted into the reflector and the excess reactivity measured. The change in reactivity with increasing temperature is the most important part of these series. The comparison between the changes with temperature for the two configurations is also essential. These experiments stand independent of all other experiments in the yttrium hydride measurement data set because they are the only ones that explicitly model the integral properties, which is an essential part of the nuclear data pipeline.

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